



materials design



WEBINAR

VASP 6: Total energies beyond DFT

Martijn Marsman
University of Vienna, Austria



Materials Design Webinar Series



► Share the webinar series with your colleagues!

- Registration details

<http://www.materialsdesign.com/webinars>

► We will be recording this webinar

- Watch any of our earlier webinars anytime
- We will post upcoming webinars on the webinar page

► Vote for the next webinar topic!

- Take a 2 minutes brief survey at the end of the webinar!



Please Ask Questions

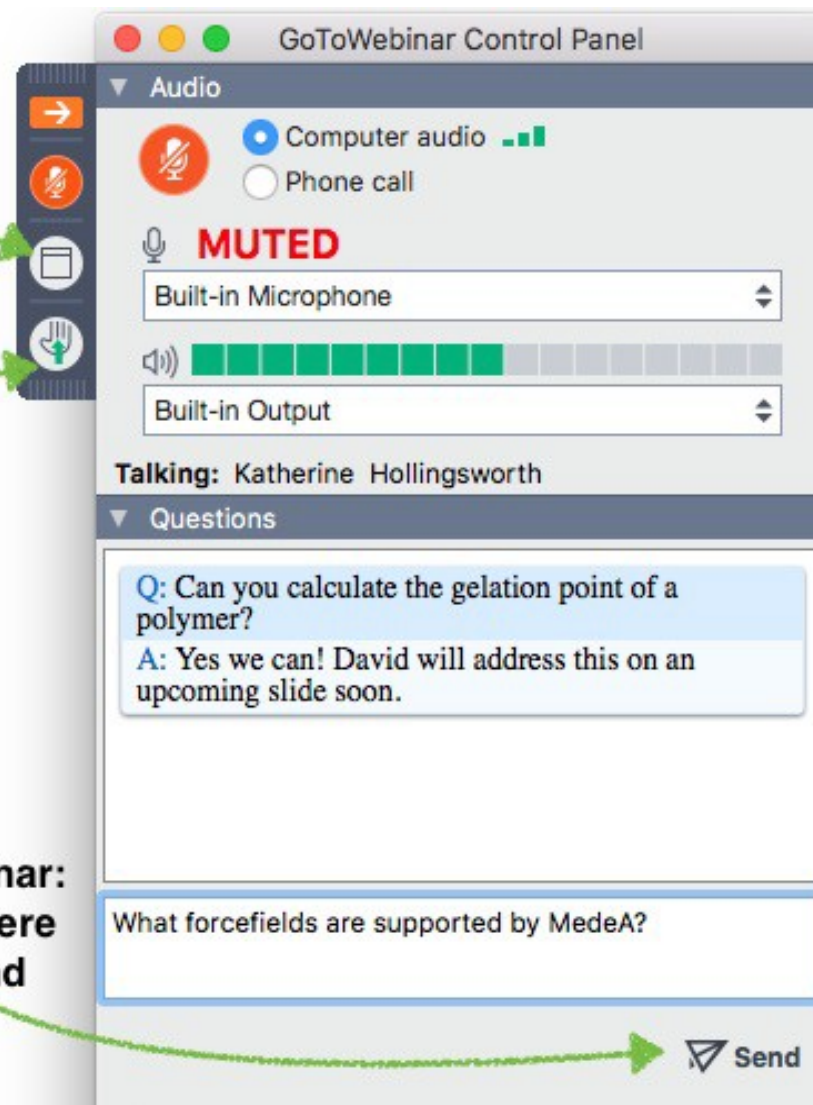


full screen

during discussion:

Use the raise hand icon to bring attention to your question

any time during webinar:
type your question here
and then press Send



Webinar Speakers



Dr. Martijn Marsman

martijn.marsman@univie.ac.at



Katherine Hollingsworth

khollingsworth@materialsdesign.com

VASP6: total energies beyond DFT

Martijn Marsman

University Vienna & VASP Software GmbH



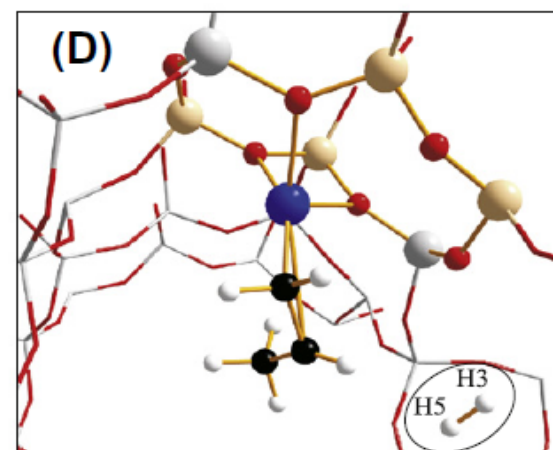
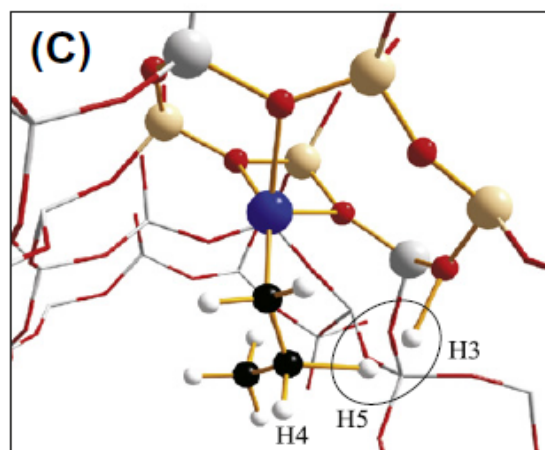
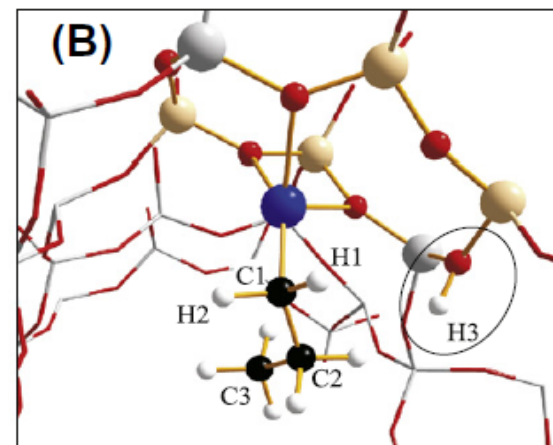
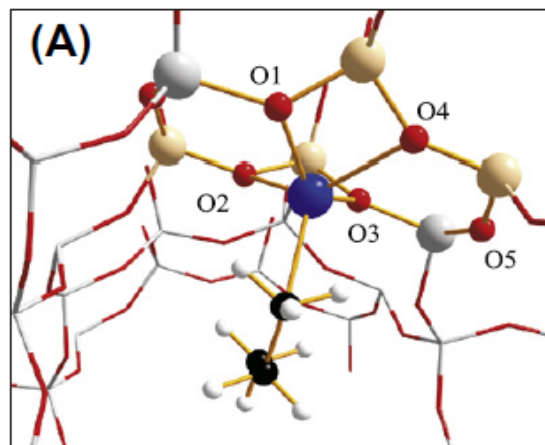
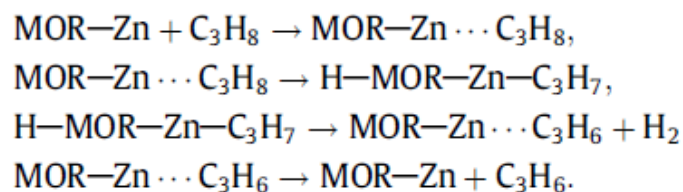
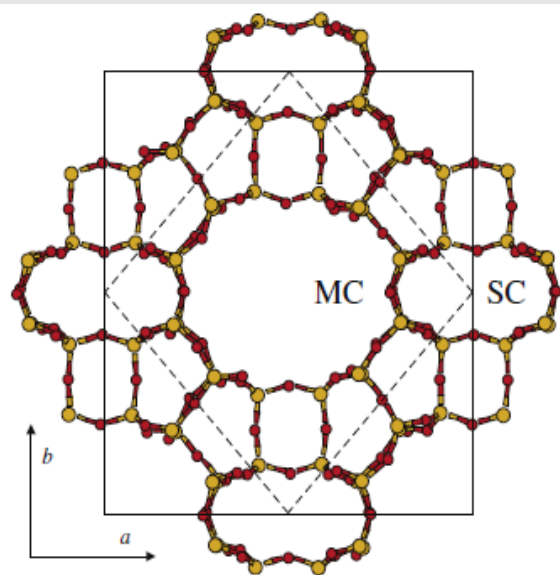
Outline

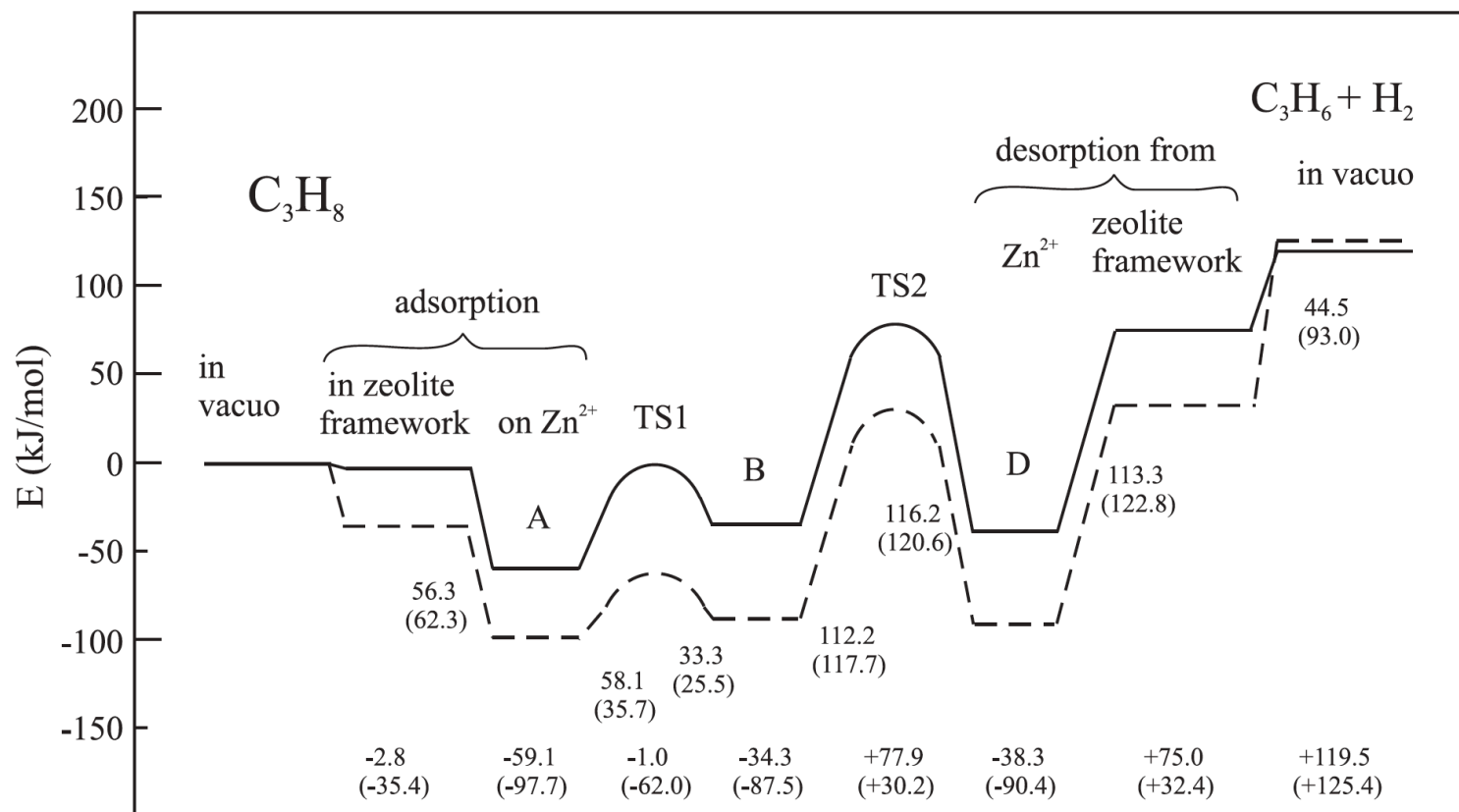
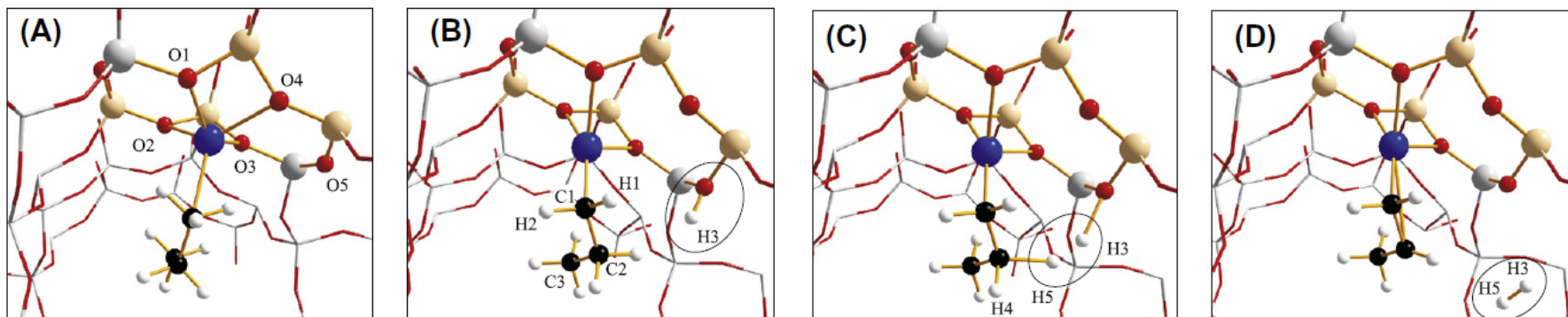
- Beyond DFT
 - The need to go beyond DFT
 - HF/DFT hybrid functionals
 - The Random-Phase-Approximation
- $O(N^3)$ implementation of the RPA
 - Cubic-scaling RPA total energies
 - Minimax frequency and time integration
 - Cubic-scaling RPA quasi-particles
- Forces in the RPA

Beyond DFT

Do we need to go beyond DFT?

Catalysis: dehydrogenation of propane in Mordenite





L. Benco *et al.*, Journal of Catalysis 227, 104 (2011)

CO adsorption on d-metal surfaces

- DFT incorrectly predicts that CO prefers the hollow site: P. Feibelman *et al.*, J. Phys. Chem. B 105, 4018 (2001)

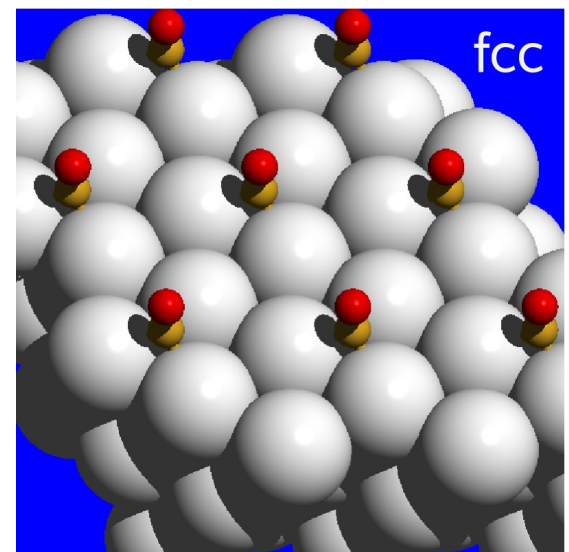
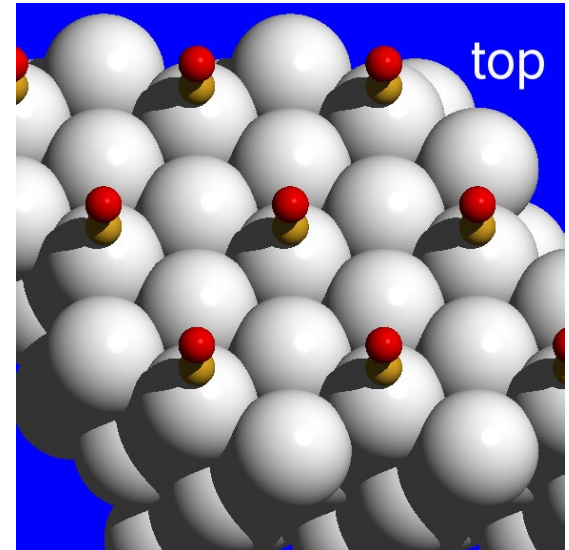
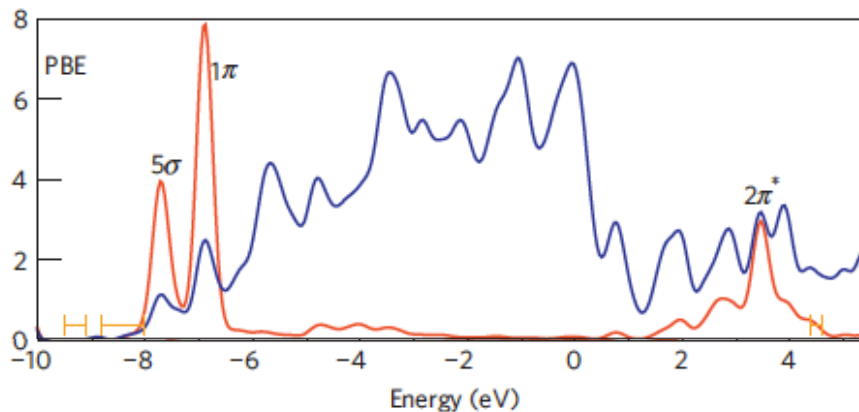
- This error is relatively large.
Best DFT/PBE calculations:

CO@Cu(111): -170 meV

CO@Rh(111): -40 meV

CO@Pt(111): -100 meV

- CO HOMO-LUMO gap too small in DFT:

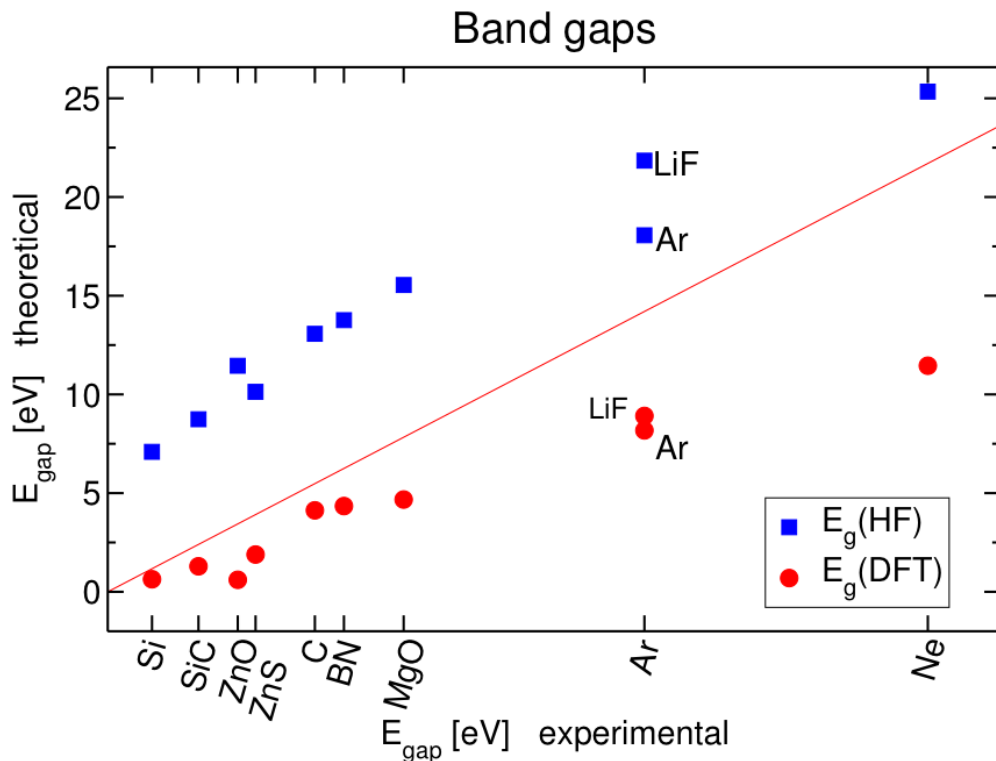


Need to go beyond DFT?

Lattice constants and Bulk moduli:
AlP, AlAs, BAs, BP, Si, C, SiC, MgO, LiF

	LDA		PBE		HF	
	Δa_0	ΔB_0	Δa_0	ΔB_0	Δa_0	ΔB_0
MRE	-1.4	3.5	0.8	-7.2	0.4	8.2
MARE	1.4	7.9	0.8	7.2	0.7	8.2

(All in %)



ΔH (kJ/mol)	PBE	EXP
$\text{Al} + \text{N}_2 \rightarrow \text{AlN}$	262	350
$\text{Mg} + \text{H}_2 \rightarrow \text{MgH}_2$	52	78
$\text{Si} + \text{C} \rightarrow \text{SiC}$	51	69
$\text{CO} \rightarrow \text{CO@Rh}$	183	144

(More) accurate treatment of electronic correlation needed for, e.g:

- Band gaps (optical properties)
- Total energy differences with **"chemical accuracy"** (1 kcal/mol \approx 40 meV):
Atomization-, formation energies, reaction barriers, *etc*
- Van der Waals interactions

Hybrid functionals

Hartree-Fock-DFT hybrid:

$$E_{\text{xc}}^{\text{hyb.}} = aE_X^{\text{HF}} + (1 - a)E_X^{\text{DFT}} + E_c^{\text{DFT}}$$

where

$$E_x^{\text{HF}} \propto \sum_{i,j}^{\text{occ.}} \int \int d^3\mathbf{r} d^3\mathbf{r}' \frac{\psi_i^*(\mathbf{r})\psi_j(\mathbf{r})\psi_j^*(\mathbf{r}')\psi_i(\mathbf{r}')}{|\mathbf{r} - \mathbf{r}'|}$$

Solve a one-electron (Roothaan) equation:

$$\left(-\frac{1}{2}\Delta + V_{\text{ext}}(\mathbf{r}) + V_{\text{H}}(\mathbf{r}) + \right)\psi_i(\mathbf{r}) + \int V_X(\mathbf{r}, \mathbf{r}')\psi_i(\mathbf{r}')d\mathbf{r}' = \epsilon_i\psi_i(\mathbf{r})$$

with an orbital dependent, non-local potential

(compare to DFT)

$$V_X(\mathbf{r}, \mathbf{r}') = - \sum_j^{\text{occ.}} \frac{\psi_j(\mathbf{r})\psi_j^*(\mathbf{r}')}{|\mathbf{r} - \mathbf{r}'|}$$

$$V_{\text{xc}}[\rho](\mathbf{r})\psi_i(\mathbf{r})$$

Hybrid functionals

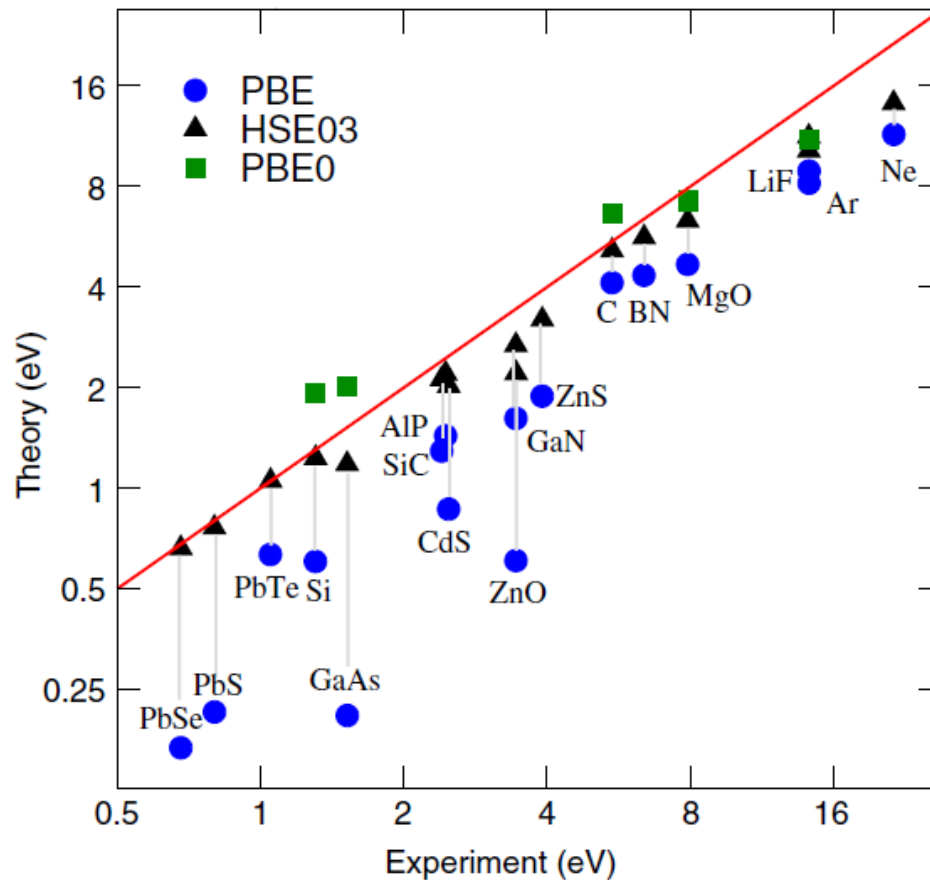


Figure 8. Band gaps from PBE, PBE0, and HSE03 calculations, plotted against data from experiment.

Lattice constant		
	MRE	MARE
PBE	0.8	1.0
PBE0	0.1	0.5
HSE	0.2	0.5
B3LYP	1.0	1.2

Bulk modulus		
	MRE	MARE
PBE	-9.8	9.4
PBE0	-1.2	5.7
HSE	-3.1	6.4
B3LYP	-10.2	11.4

Atomization energy		
	MRE	MARE
PBE	-1.9	3.4
PBE0	-6.5	7.4
HSE	-5.1	6.3
B3LYP	-17.6	17.6

(All in %)

CO adsorption on d-metal surfaces

CO @		top	fcc	hcp	Δ
Cu(111)	PBE	0.709	0.874	0.862	-0.165
	PBE0	0.606	0.579	0.565	0.027
	HSE03	0.561	0.555	0.535	0.006
	exp.	0.46-0.52			
Rh(111)	PBE	1.870	1.906	1.969	-0.099
	PBE0	2.109	2.024	2.104	0.005
	HSE03	2.012	1.913	1.996	0.016
	exp.	1.43-1.65			
Pt(111)	PBE	1.659	1.816	1.750	-0.157
	PBE0	1.941	1.997	1.944	-0.056
	HSE03	1.793	1.862	1.808	-0.069
	exp.	1.43-1.71			

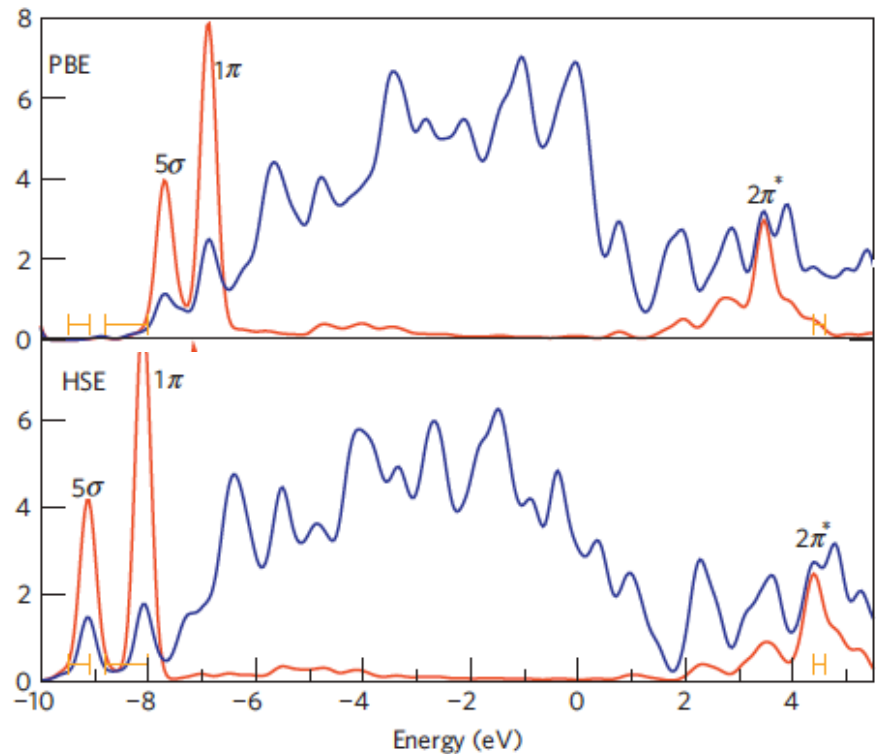
(All energies in eV)

A. Stroppa *et al.*, PRB 76, 195440 (2007); A. Stroppa and G. Kresse, NJP 10, 063020 (2008).

Hybrid functionals reduce the tendency to stabilize adsorption at the hollow site w.r.t. the top site.

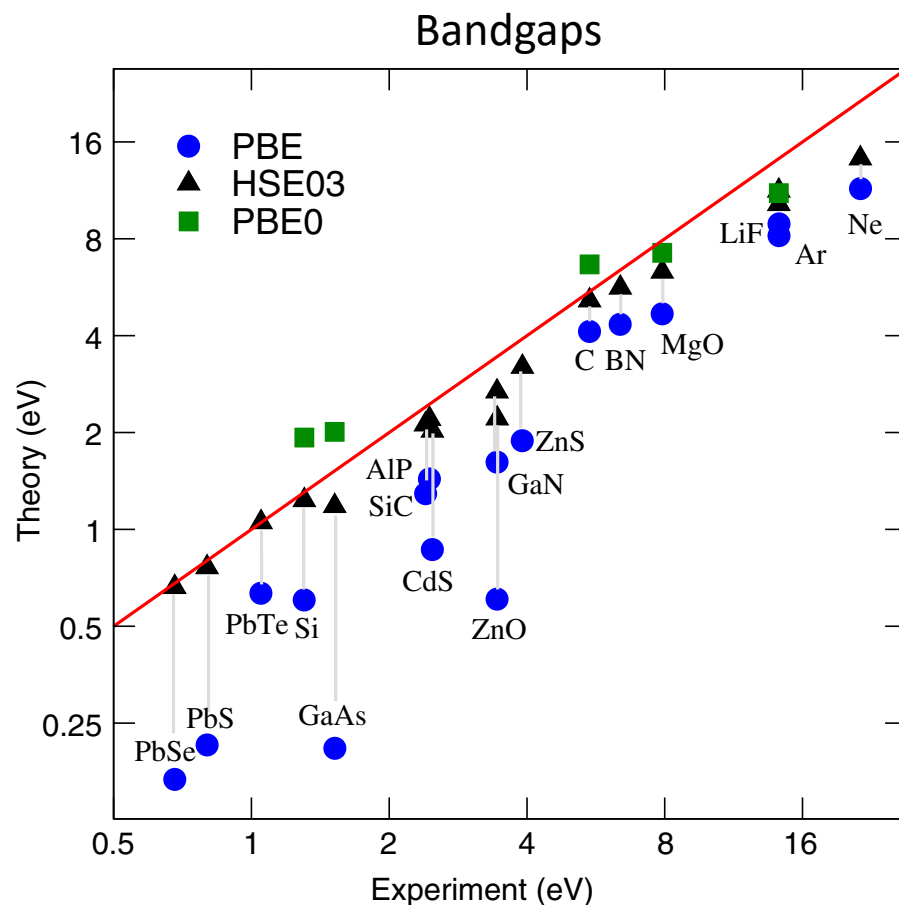
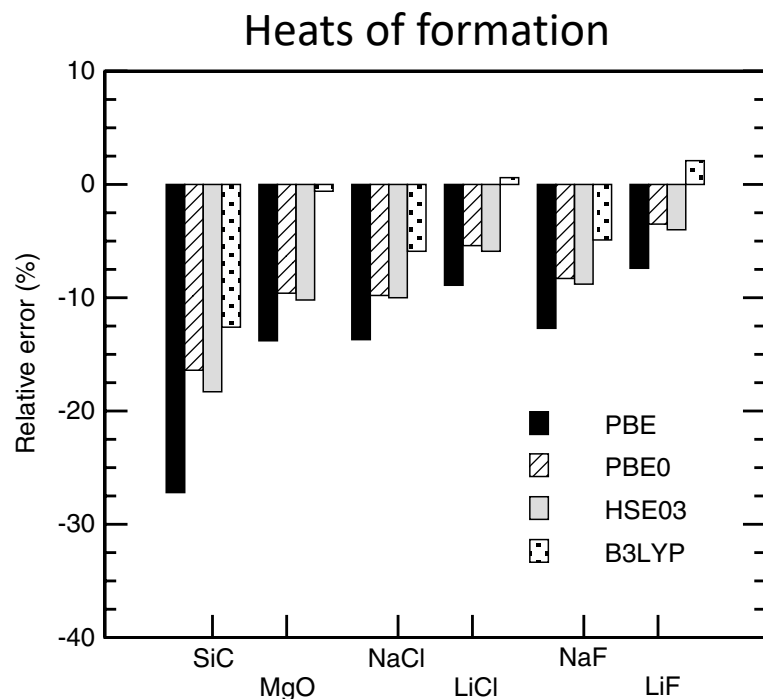
reduced CO $2\pi^*$ – metal- d interaction

- DFT does well for the metallic surface, but not for the CO: $2\pi^*$ (LUMO) too close to the Fermi level.
- HSE does well for the CO, but not for the surface: d -metal bandwidth too large.



Schimka et al., Nature Materials 9, 741 (2010)

Need to go beyond DFT and DFT/HF hybrids?



- Total energy differences with "chemical accuracy" (1 kcal/mol \approx 40 meV/atom): Atomization-, formation energies, reaction barriers, etc
- Van der Waals interactions

- Bandstructure of metals and *largish* gap systems, and some problematic cases inbetween

The Random-Phase-Approximation: GW and ACFDT

One-electron/QP picture

DFT: Kohn-Sham eq.

$$\left(-\frac{1}{2}\Delta + V_{\text{ext}}(\mathbf{r}) + V_{\text{H}}(\mathbf{r}) + V_{\text{xc}}[\rho](\mathbf{r})\right)\psi_{n\mathbf{k}}(\mathbf{r}) = \epsilon_{n\mathbf{k}}\psi_{n\mathbf{k}}(\mathbf{r})$$

DFT-HF hybrid functionals: Roothaan eq.

$$\left(-\frac{1}{2}\Delta + V_{\text{ext}}(\mathbf{r}) + V_{\text{H}}(\mathbf{r})\right)\psi_{n\mathbf{k}}(\mathbf{r}) + \int V_{\text{X}}[\{\psi_o\}](\mathbf{r}, \mathbf{r}')\psi_{n\mathbf{k}}(\mathbf{r}')d\mathbf{r}' = \epsilon_{n\mathbf{k}}\psi_{n\mathbf{k}}(\mathbf{r})$$

GW: quasi-particle eq.

$$\left(-\frac{1}{2}\Delta + V_{\text{ext}}(\mathbf{r}) + V_{\text{H}}(\mathbf{r})\right)\psi_{n\mathbf{k}}(\mathbf{r}) + \int \Sigma[\{\psi, E\}](\mathbf{r}, \mathbf{r}', E_{n\mathbf{k}})\psi_{n\mathbf{k}}(\mathbf{r}')d\mathbf{r}' = E_{n\mathbf{k}}\psi_{n\mathbf{k}}(\mathbf{r})$$

GW


The quasi-particle equation:

$$\left(-\frac{1}{2}\Delta + V_{\text{ext}}(\mathbf{r}) + V_{\text{H}}(\mathbf{r})\right) \psi_{n\mathbf{k}}(\mathbf{r}) + \int \Sigma(\mathbf{r}, \mathbf{r}', E_{n\mathbf{k}}) \psi_{n\mathbf{k}}(\mathbf{r}') d\mathbf{r}' = E_{n\mathbf{k}} \psi_{n\mathbf{k}}(\mathbf{r})$$


The “self-energy is given by: $\Sigma = iGW$

or more explicitly

$$\Sigma(\mathbf{r}, \mathbf{r}', E) = \frac{i}{2\pi} \int_{-\infty}^{\infty} d\omega \sum_n^{\text{all}} \frac{\psi_n(\mathbf{r}) \psi_n^*(\mathbf{r}')}{\omega - E - E_n + i\eta \text{sgn}(E_n - E_{\text{fermi}})} \times$$


Green's function: G 

$$\times e^2 \int d\mathbf{r}'' \frac{\epsilon^{-1}(\mathbf{r}, \mathbf{r}'', \omega)}{|\mathbf{r}'' - \mathbf{r}'|}$$

screened Coulomb interaction: W 

Compare to Fock-exchange:

$$V_{\text{X}}(\mathbf{r}, \mathbf{r}') = - \sum_n^{\text{occ.}} \psi_n(\mathbf{r}) \psi_n^*(\mathbf{r}') \times \frac{e^2}{|\mathbf{r} - \mathbf{r}'|}$$

bare Coulomb interaction: v 

The Green's function: physical interpretation

The Green's function $G(\mathbf{r}, \mathbf{r}', t - t')$ describes the propagation of a particle from (\mathbf{r}, t) to (\mathbf{r}', t') : i.e., provided we have particle at position \mathbf{r} at time t , $G(\mathbf{r}, \mathbf{r}', t - t')$ is the chance of finding it at position \mathbf{r}' at time t' .

$$G_0(\mathbf{r}, \mathbf{r}', \omega) = \sum_n^{\text{all}} \frac{\psi_n^*(\mathbf{r}) \psi_n(\mathbf{r}')}{\omega - \epsilon_n + i\eta \operatorname{sgn}(\epsilon_n - \mu)}$$

- Particle propagator: $G_o(1,2) = G_o(\mathbf{r}_1, \mathbf{r}_2, t_2 - t_1)$ for $t_2 > t_1$:

$$(\mathbf{r}_1, t_1) = 1 \quad (\mathbf{r}_2, t_2) = 2$$



$$G_0(1, 2) = \sum_n^{\text{vir.}} \psi_n(\mathbf{r}_1)^* \psi_n(\mathbf{r}_2) e^{-i(\epsilon_n - \mu)(t_2 - t_1)}$$

- Hole propagator: $G_o(1,2)$ for $t_1 > t_2$:

$$(\mathbf{r}_2, t_2) = 2 \quad (\mathbf{r}_1, t_1) = 1$$



$$G_0(1, 2) = \sum_n^{\text{occ.}} \psi_n^*(\mathbf{r}_1) \psi_n(\mathbf{r}_2) e^{-i(\epsilon_n - \mu)(t_1 - t_2)}$$

W in the Random-Phase-Approximation

Key: the “irreducible polarizability in the independent particle picture” χ^0 (or χ^{KS}):

$$\chi^0(\mathbf{r}, \mathbf{r}', \omega) := \frac{\partial \rho_{\text{ind}}(\mathbf{r}, \omega)}{\partial v_{\text{eff}}(\mathbf{r}', \omega)}$$

In the RPA, the screened Coulomb interaction is computed from χ^0 as:

$$W = \nu + \nu\chi_0\nu + \nu\chi_0\nu\chi_0\nu + \nu\chi_0\nu\chi_0\nu\chi_0\nu + \dots = \nu \underbrace{(1 - \chi_0\nu)^{-1}}_{\epsilon^{-1}}$$

1. The bare Coulomb interaction between two particles

2. The electronic environment reacts to the field generated by a particle: induced change in the density $\chi_0\nu$, that gives rise to a change in the Hartree potential: $\nu\chi_0\nu$.

3. The electrons react to the induced change in the potential: additional change in the density, $\chi_0\nu\chi_0\nu$, and corresponding change in the Hartree potential: $\nu\chi_0\nu\chi_0\nu$.

and so on, and so on ...

geometrical series

$$W = \epsilon^{-1}\nu$$

$$\epsilon^{-1} = 1 + \nu\chi$$

$$\chi = \chi^0 + \chi^0\nu\chi$$

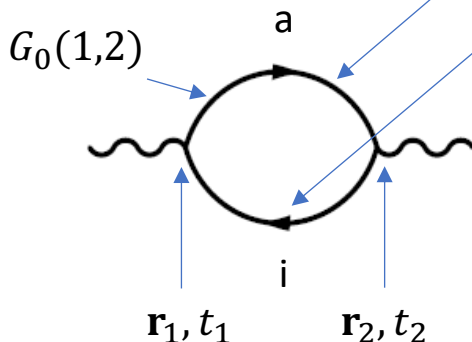
The IP-polarizability: χ_0

The “irreducible polarizability in the independent particle picture” χ^0 (or χ^{KS}):

$$\chi^0(\mathbf{r}, \mathbf{r}', \omega) := \frac{\partial \rho_{\text{ind}}(\mathbf{r}, \omega)}{\partial v_{\text{eff}}(\mathbf{r}', \omega)}$$

Adler and Wiser derived expressions for χ^0

$$\chi_0(\mathbf{r}_1, \mathbf{r}_2, \omega) = \sum_i^{\text{occ.}} \sum_a^{\text{virt.}} \frac{\langle \psi_a | \mathbf{r}_1 | \psi_i \rangle \langle \psi_i | \mathbf{r}_2 | \psi_a \rangle}{\epsilon_i - \epsilon_a - \omega} + \sum_i^{\text{occ.}} \sum_a^{\text{virt.}} \frac{\langle \psi_i | \mathbf{r}_1 | \psi_a \rangle \langle \psi_a | \mathbf{r}_2 | \psi_i \rangle}{\epsilon_a - \epsilon_i - \omega}$$

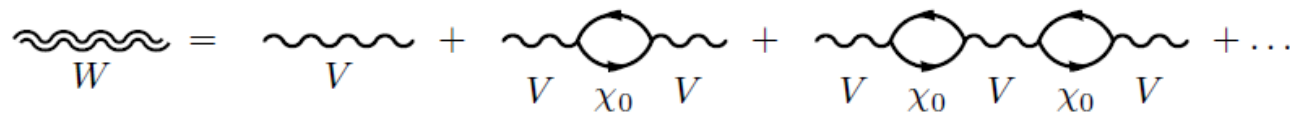
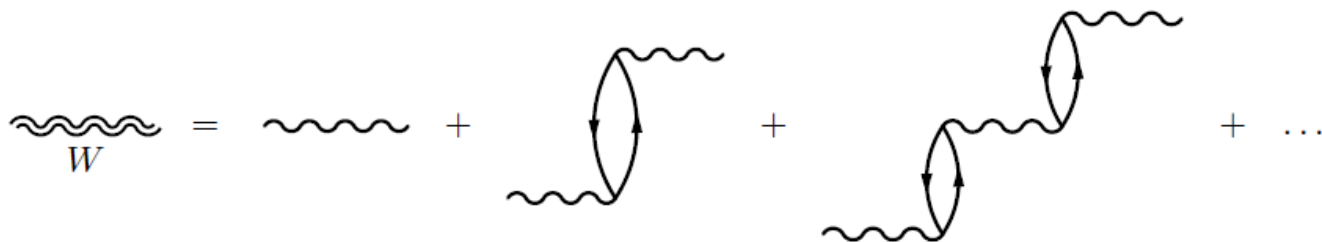
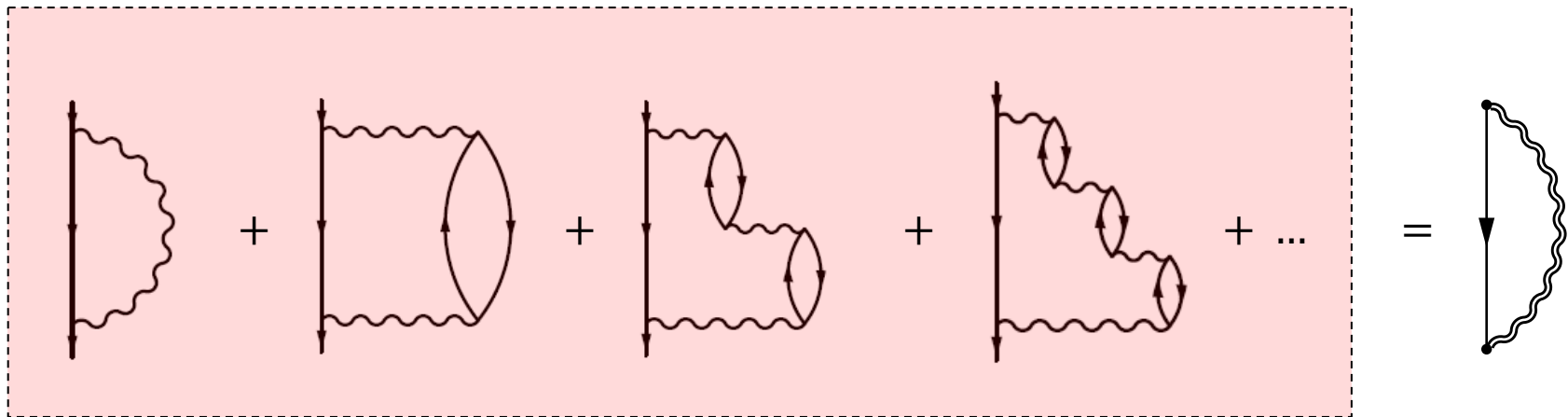


Or in terms of Green's functions (propagators):

$$\chi_0(\mathbf{r}_1, t_1, \mathbf{r}_2, t_2) = \chi(1, 2) = -G_0(1, 2)G_0(2, 1)$$

$$W = V + V \chi_0 V + V \chi_0 V \chi_0 V + \dots$$

GW in the RPA



RPA total energies (ACFDT)

The “RPA” total energy is given by:

$$E[n] = T_{KS}[\{\psi_i\}] + E_H[n] + E_X[\{\psi_i\}] + E_{\text{ion-el}}[n] + E_c$$

with the RPA correlation energy

$$E_c = \sum_{\mathbf{q}} \int_0^\infty \frac{d\omega}{2\pi} \text{Tr}\{\ln[1 - \chi_0(\mathbf{q}, i\omega)\nu] + \chi_0(\mathbf{q}, i\omega)\nu\}$$

and the Independent-Particle polarizability: $\chi^0(\mathbf{r}, \mathbf{r}', \omega) := \frac{\partial \rho_{\text{ind}}(\mathbf{r}, \omega)}{\partial v_{\text{eff}}(\mathbf{r}', \omega)}$

$$\chi_{\mathbf{G}, \mathbf{G}'}^0(\mathbf{q}, \omega) = \frac{1}{\Omega} \sum_{nn'\mathbf{k}} 2w_{\mathbf{k}} (f_{n'\mathbf{k}+\mathbf{q}} - f_{n'\mathbf{k}}) \\ \times \frac{\langle \psi_{n'\mathbf{k}+\mathbf{q}} | e^{i(\mathbf{q}+\mathbf{G})\mathbf{r}} | \psi_{n\mathbf{k}} \rangle \langle \psi_{n\mathbf{k}} | e^{-i(\mathbf{q}+\mathbf{G}')\mathbf{r}'} | \psi_{n'\mathbf{k}+\mathbf{q}} \rangle}{\epsilon_{n'\mathbf{k}+\mathbf{q}} - \epsilon_{n\mathbf{k}} - \omega - i\eta}$$

$$\ln(1 - \chi_0\nu) + \chi_0\nu = -\frac{\text{Tr}[\chi_0\nu\chi_0\nu]}{2} - \frac{\text{Tr}[\chi_0\nu\chi_0\nu\chi_0\nu]}{3} - \frac{\text{Tr}[\chi_0\nu\chi_0\nu\chi_0\nu\chi_0\nu]}{4} \dots$$

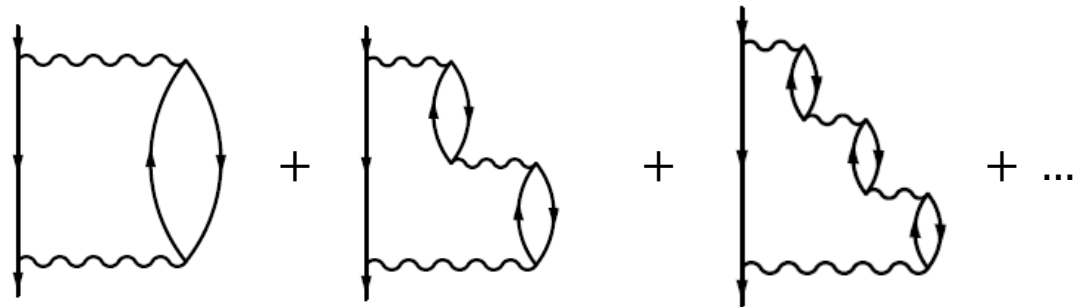
In terms of diagrams

$$\ln(1 - \chi_0 v) + \chi_0 v = -\frac{\text{Tr}[\chi_0 v \chi_0 v]}{2} - \frac{\text{Tr}[\chi_0 v \chi_0 v \chi_0 v]}{3} - \frac{\text{Tr}[\chi_0 v \chi_0 v \chi_0 v \chi_0 v]}{4} + \dots$$

2nd. order 3rd. order 4th. order

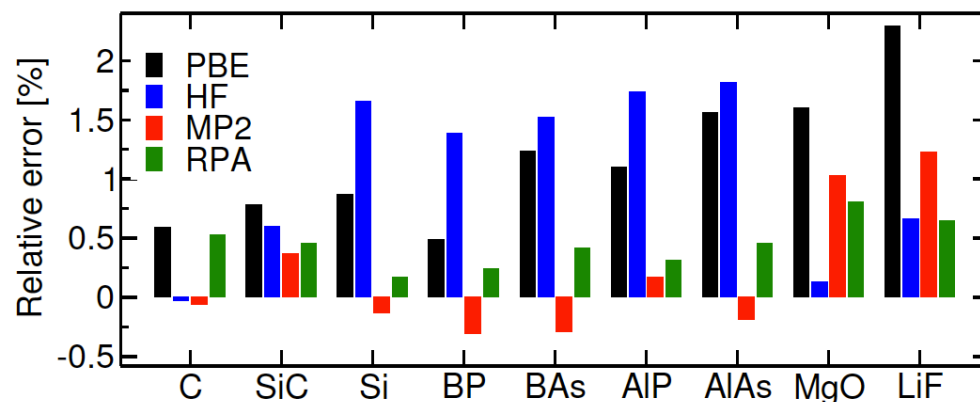
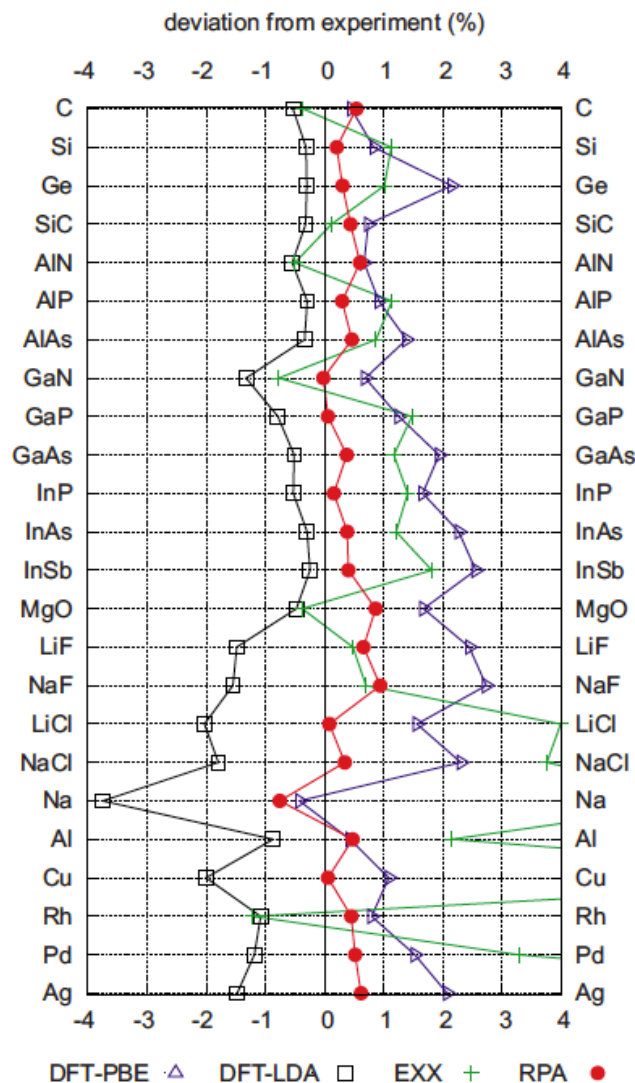
The derivate w.r.t. G
yields GW!

$$\frac{\partial E_c}{\partial G} = \Sigma$$



RPA: lattice constants

J. Harl et al., PRB 81, 115126 (2010)



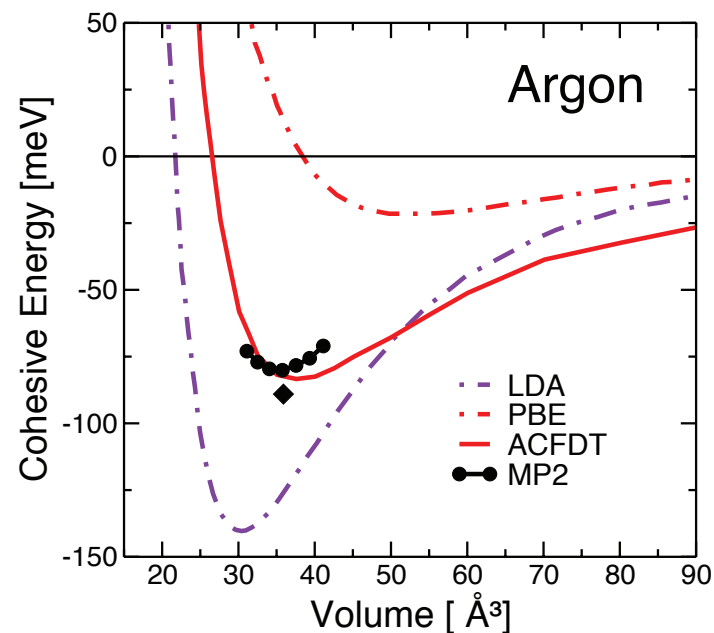
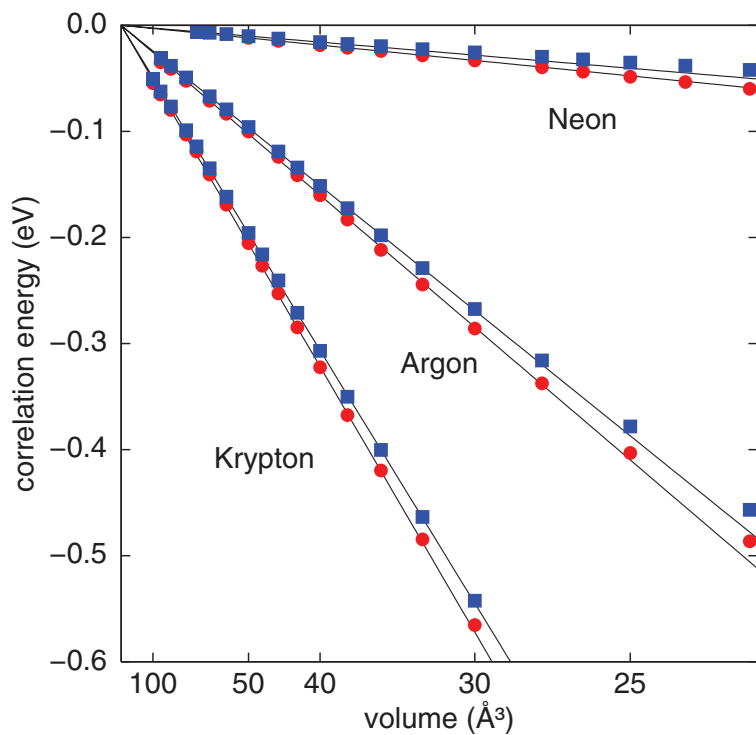
Deviations w.r.t. experiment
(corrected for zero-point vibrations)

	MRE	MARE
PBE	1.2	1.2
HF	1.1	1.1
MP2	0.2	0.4
RPA	0.5	0.4

(in %)

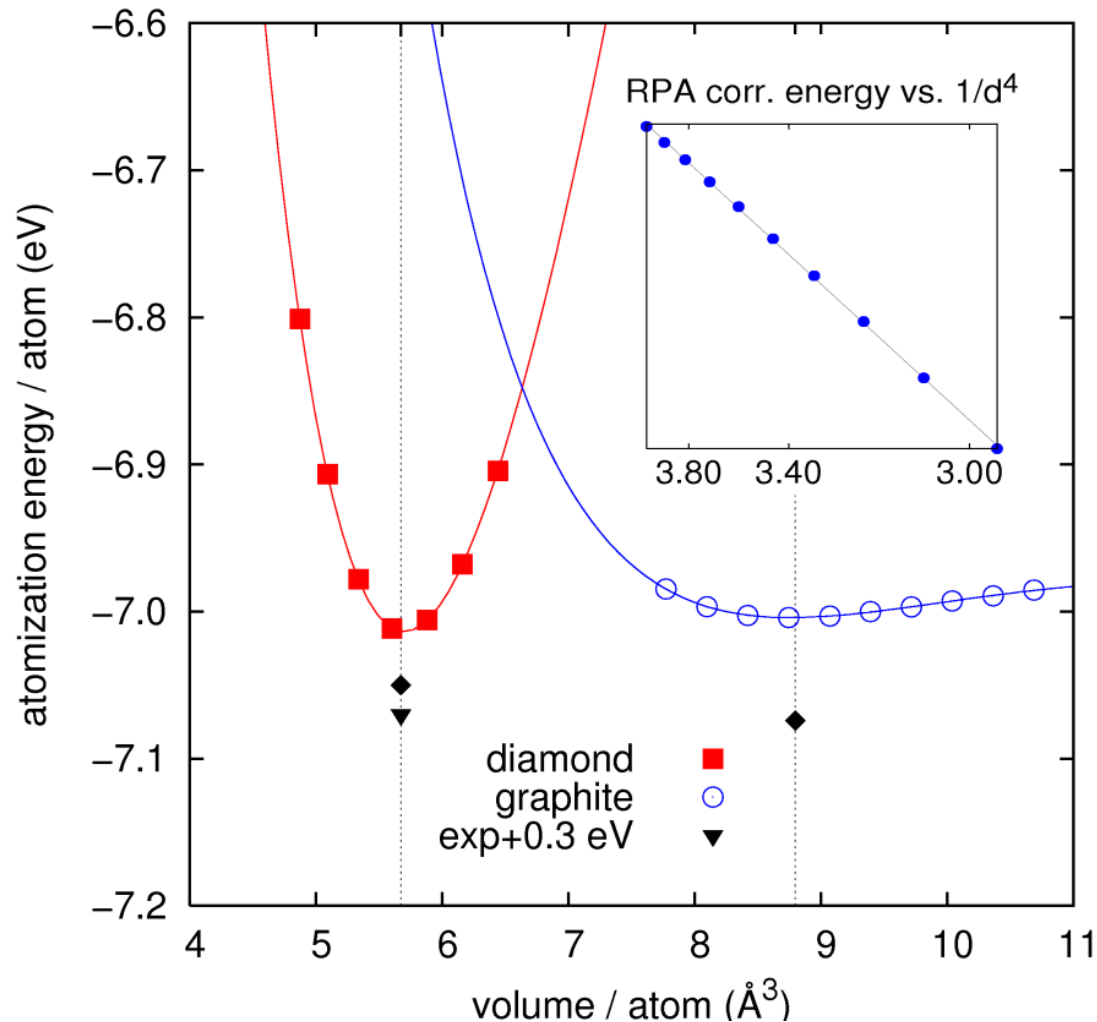
RPA: noble gas solids

J. Harl and G. Kresse, PRB 77, 045136 (2008)



	C_6 coefficients		
	RPA(LDA)	RPA(PBE)	Exp.
Ne	62	53	47
Ar	512	484	455
Kr	1030	980	895

Graphite vs. Diamond



$1/d^4$ behavior at short distances

	QMC (Galli)	RPA	EXP
$d(\text{\AA})$	3.426	3.34	3.34
C_{33}		36	36-40
$E(\text{meV})$	56	48	43-50

J. Harl, G. Kresse,
PRL 103, 056401 (2009).
S. Lebeque, et al.,
PRL 105, 196401 (2010).

RPA: heats of formation

J. Harl and G. Kresse, PRL 103, 056401 (2009)

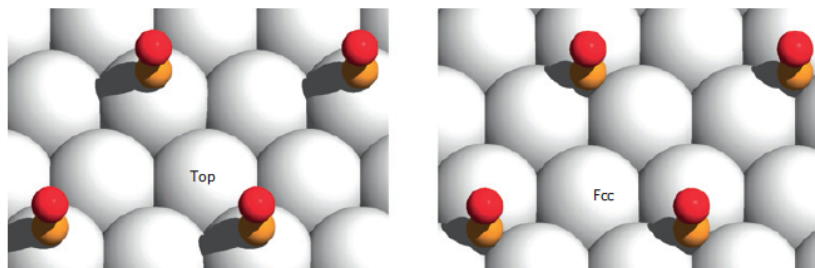
Heats of formation w.r.t. normal state at ambient conditions (in kJ/mol)

Example: $\text{Mg}(\text{bulk metal}) + \text{H}_2 \rightarrow \text{MgH}_2$

	PBE	Hartree-Fock	RPA	EXP
LiF	570	664	609	621
NaF	522	607	567	576
NaCl	355	433	405	413
MgO	516	587	577	603
MgH ₂	52	113	72	78
AlN	262	350	291	321
SiC	51	69	64	69

RPA: CO @ Pt(111) and Rh(111)

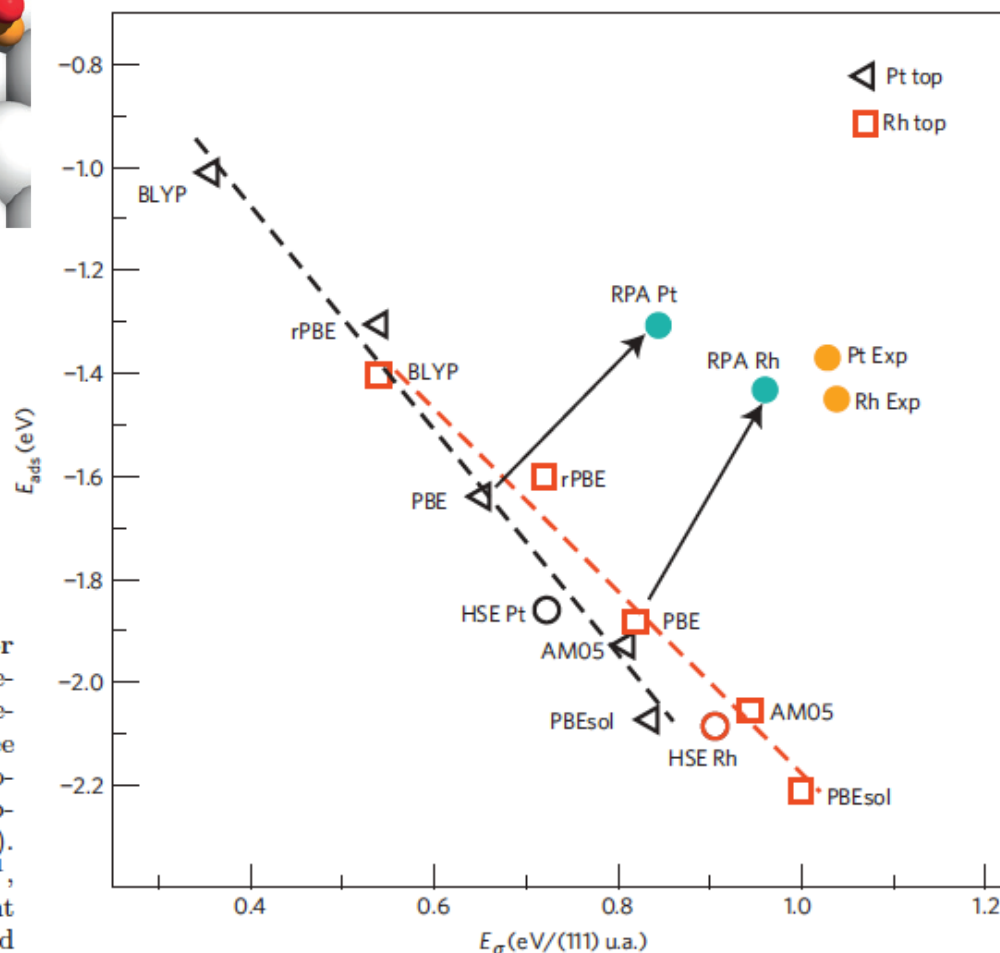
Schimka et al., Nature Materials 9, 741 (2010)



RPA:

- increases surface energy and
- decreases adsorption energy

FIG. 1: Atop CO adsorption and surface energies for Pt(111) and Rh(111). (a) Considered CO adsorption geometries for a (2×2) surface cell. Semi-local functionals predict CO to adsorb in the fcc hollow site coordinated to three metal atoms on Pt and Rh, whereas experiments unequivocally show adsorption atop a metal atom. (b) Atop adsorption energies versus surface energies for Pt(111) and Rh(111). Various semi-local functionals were used: AM05¹⁰, PBEsol¹¹, PBE⁸, rPBE¹² and BLYP¹³, in order of increasing gradient corrections. Furthermore the hybrid functional HSE¹⁸ based on the PBE functional was used.



- DFT does well for the metallic surface, but not for the CO: $2\pi^*$ (LUMO) too close to the Fermi level.
- HSE does well for the CO, but not for the surface: d -metal bandwidth too large.
- GW gives a good description of both the metallic surface as well as of the CO $2\pi^*$ (LUMO). The CO 5σ and 1π are slightly underbound.

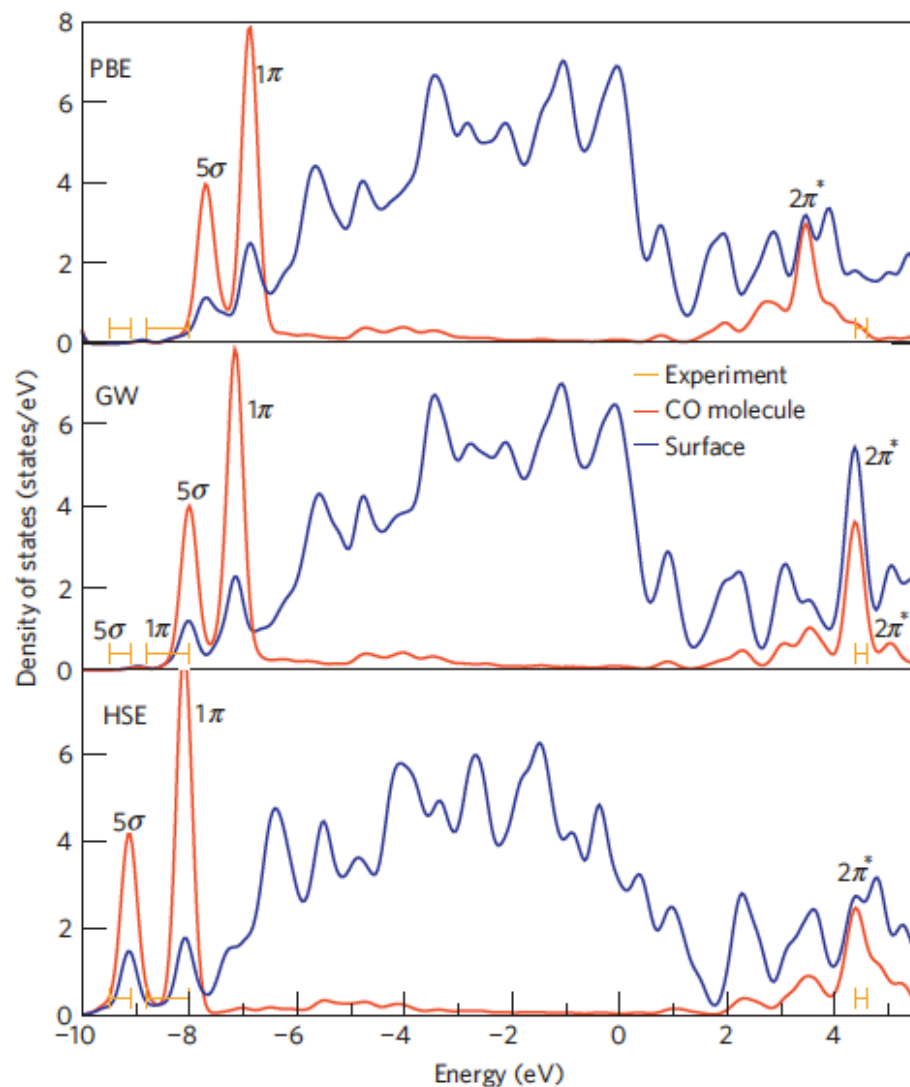
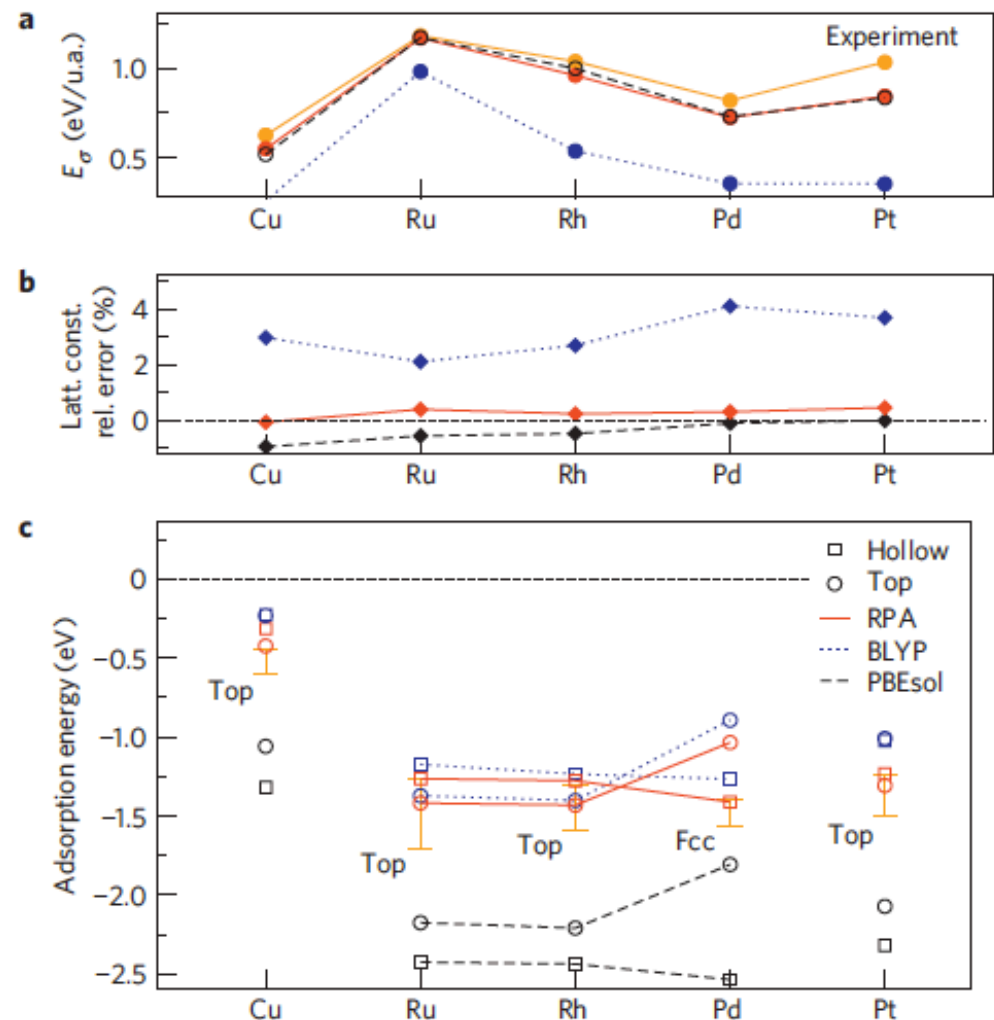


Figure 2 | Electronic DOS for CO adsorbed atop a Pt atom on Pt(111). The DOS is evaluated using DFT (PBE), the RPA (GW) and a hybrid functional (HSE). Experimental photoemission data for the $2\pi^*$ state are from ref. 19, for the 5σ and 1π state from ref. 20.

RPA:

- Right site preference
- Good adsorption energies
- Excellent lattice constants
- Good surface energies

Figure 3 | Surface energies, lattice constants and adsorption energies. **a**, Fcc(111) surface energies (E_σ) for PBEsol, BLYP and RPA. Experimental surface energies are deduced from liquid-metal data^{24,25}. **b**, Lattice constants for PBEsol, RPA and BLYP. **c**, Adsorption energies for the atop and hollow sites of CO on Cu, late 4d metals and Pt for PBEsol, RPA and BLYP. Experimental data with error bars are from ref. 26. The error bars correspond to the spread of the experimental results.



Cubic scaling RPA

As shown by Adler and Wiser, the IP polarizability χ^0 can be straightforwardly calculated from $\{\psi, \epsilon\}$:

$$\chi_{\mathbf{G}, \mathbf{G}'}^0(\mathbf{q}, \omega) = \frac{1}{\Omega} \sum_{nn'\mathbf{k}} 2w_{\mathbf{k}} (f_{n'\mathbf{k}+\mathbf{q}} - f_{n'\mathbf{k}}) \\ \times \frac{\langle \psi_{n'\mathbf{k}+\mathbf{q}} | e^{i(\mathbf{q}+\mathbf{G})\mathbf{r}} | \psi_{n\mathbf{k}} \rangle \langle \psi_{n\mathbf{k}} | e^{-i(\mathbf{q}+\mathbf{G}')\mathbf{r}'} | \psi_{n'\mathbf{k}+\mathbf{q}} \rangle}{\epsilon_{n'\mathbf{k}+\mathbf{q}} - \epsilon_{n\mathbf{k}} - \omega - i\eta}$$

Expensive: this scales as N^4 !

Cubic-scaling RPA

M. Kaltak, J. Klimeš, and G. Kresse, PRB 90, 054115 (2014)

Evaluate the Green's function in “imaginary” time:

$$G(\mathbf{r}, \mathbf{r}', i\tau) = \sum_n \psi_n(\mathbf{r}) \psi_n^*(\mathbf{r}') e^{-\epsilon_n \tau}$$

and the polarizability as:

$$\chi^0(\mathbf{r}, \mathbf{r}', i\tau) = -G(\mathbf{r}, \mathbf{r}', i\tau) G(\mathbf{r}, \mathbf{r}', -i\tau)$$

Followed by a cosine-transform:

$$\chi^0(\mathbf{r}, \mathbf{r}', i\tau) \xrightarrow{\text{CT}} \chi^0(\mathbf{r}, \mathbf{r}', i\omega)$$

Now the worst scaling steps are the evaluation of the Greens function (N^3), and

$$E_c = \int_0^\infty \frac{d\omega}{2\pi} \text{Tr} \{ \ln[1 - \chi_0(i\omega)\nu] + \chi_0(i\omega)\nu \}$$

(which scales as N^3 due to the diagonalization involved in evaluating the “ln”).

The evaluation of the polarizability scales quadratically (N^2)!

But storing G and χ is expensive! → we need small sets of cleverly chosen “ τ ” and “ ω ”

The “minimax” fit

Express a function $f(x)$ in a basis of N functions $\{\varphi(\alpha_i, x) | i = 1, \dots, N\}$, such that the error:

$$\eta(\vec{\alpha}, \vec{\beta}, x) = f(x) - \sum_i \beta_i \varphi(\alpha_i, x)$$

is “as small as possible”.

Choose $\{(\alpha_i, \beta_i) | i = 1, \dots, N\}$ so that the *maximum error*:

$$\|\eta\|_\infty := \max_{a \leq x \leq b} |\eta(\vec{\alpha}, \vec{\beta}, x)|$$

is minimized.

This solution, $\{(\alpha_i^*, \beta_i^*) | i = 1, \dots, N\}$, is the so-called “minimax” solution

The “minimax” frequency grid

M. Kaltak, J. Klimeš, and G. Kresse, JCTC 10, 2498 (2014)

The IP polarizability can be written as (Adler&Wiser):

$$\chi(i\omega) = \sum_{\mu} \chi_{\mu} \phi(\omega, x_{\mu})$$

$$\chi_{\mu(=ia)} = \langle i | \mathbf{r} | a \rangle \langle a | \mathbf{r}' | i \rangle$$

$$x_{\mu(=ia)} = \epsilon_a - \epsilon_i$$

where $\phi(\omega, x) = \frac{2x}{x^2 + \omega^2}$

Determine a frequency grid and quadrature weights by expressing the “2nd-order direct Møller-Plesset energy”

$$E_c^{(2)} = -\frac{1}{8\pi} \int_{-\infty}^{\infty} d\omega \{ \chi(i\omega) \nu \}^2$$

in the basis $\{ \phi(\omega_i, x) | i = 1, \dots, N \}$ in the minimax sense:

$$\{ (\gamma_k^*, \omega_k^*) | k = 1, \dots, N \} \quad \eta(\vec{\gamma}, \vec{\omega}, x) = \frac{1}{x} - \frac{1}{\pi} \sum_{k=1}^N \gamma_k \phi^2(\omega_k, x) \quad E_g \leq x \leq \max(\epsilon_a - \epsilon_i)$$

$$E_c^{(2)} = -\frac{1}{8\pi} \int_{-\infty}^{\infty} d\omega \{ \chi(i\omega) \nu \}^2 \longrightarrow E_c^{(2)} \approx -\frac{1}{8\pi} \sum_{k=1}^N \gamma_k^* \{ \chi(i\omega_k^*) \nu \}^2$$

The “minimax” time grid

M. Kaltak, J. Klimeš, and G. Kresse, JCTC 10, 2498 (2014)

Alternatively the IP polarizability can be written as:

$$\hat{\chi}(i\tau) = \sum_{\mu} \chi_{\mu} \hat{\phi}(\tau, x_{\mu}) \quad \chi_{\mu(=ia)} = \langle i | \mathbf{r} | a \rangle \langle a | \mathbf{r}' | i \rangle$$
$$x_{\mu(=ia)} = \epsilon_a - \epsilon_i$$

where $\hat{\phi}(\tau, x) = e^{-x|\tau|}$

Determine a frequency grid and quadrature weights by expressing the “2nd-order direct Møller-Plesset energy”

$$E_c^{(2)} = -\frac{1}{4} \int_{-\infty}^{\infty} d\tau \{ \hat{\chi}(i\tau) \nu \}^2$$

in the basis $\{ \hat{\phi}(\tau_i, x) | i = 1, \dots, N \}$ in the minimax sense:

$$\{ (\sigma_k^*, \tau_k^*) | k = 1, \dots, N \} \quad \hat{\eta}(\vec{\sigma}, \vec{\tau}, x) = \frac{1}{2x} - \sum_{k=1}^N \sigma_k \hat{\phi}^2(\tau_k, x) \quad E_g \leq x \leq \max(\epsilon_a - \epsilon_i)$$

$$E_c^{(2)} = -\frac{1}{8\pi} \int_{-\infty}^{\infty} d\omega \{ \chi(i\omega) \nu \}^2 \longrightarrow E_c^{(2)} \approx -\frac{1}{4} \sum_{k=1}^N \sigma_k^* \{ \hat{\chi}(i\tau_k^*) \nu \}^2$$

The non-uniform cosine transform

M. Kaltak, J. Klimeš, and G. Kresse, JCTC 10, 2498 (2014)

The cosine transform of $\chi(i\omega) \leftrightarrow \chi(i\tau)$:

$$\chi(i\omega) = 2 \int_0^\infty d\tau \hat{\chi}(i\tau) \cos(\omega\tau) \qquad \hat{\chi}(i\tau) = \frac{1}{\pi} \int_0^\infty d\omega \chi(i\omega) \cos(\omega\tau)$$

is reformulated on the minimax frequency and time grids as

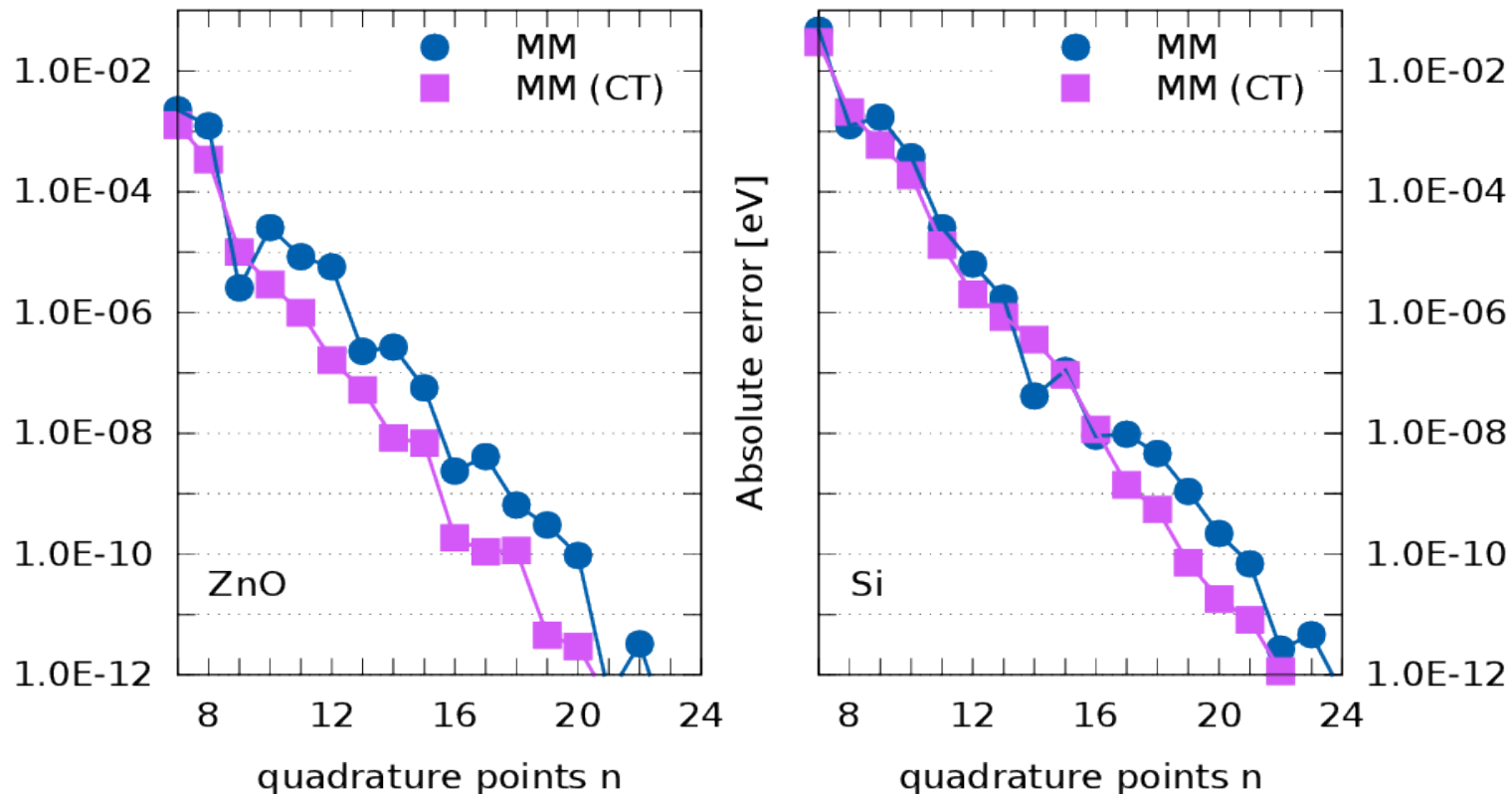
$$\chi(i\omega_k) = \sum_{l=1}^N a_{kl} \hat{\chi}(i\tau_l) \qquad a_{kl} = w_{kl} \cos(\omega_k^* \tau_l^*)$$

where the coefficients w_{kl} are the minimax solution to the following fitting problem:

$$\eta^c(\vec{w}_k, x) = \phi(\omega_k^*, x) - \sum_{l=1}^N w_{kl} \cos(\omega_k^* \tau_l^*) \hat{\phi}(\tau_l^*, x) \xrightarrow{\text{minimax fit}} \{w_{kl}^* | k, l = 1, \dots, N\}$$

for fixed $\{\omega_k^*\}$ and $\{\tau_k^*\}$, and for $E_g \leq x \leq \max(\epsilon_a - \epsilon_i)$.

How good are the grids: ZnO and Si



- N^4 RPA calculations entirely in frequency
- N^2 calculation of polarizability in time, then transformation to frequency (200 atoms on 200 cores in about one hour)

M. Kaltak, J. Klimeš, and G. Kresse, JCTC 10, 2498 (2014); PRB 90, 054115 (2014).

Scaling

New RPA code (VASP6):

- Scales linearly in the number of k-points (as DFT), instead of quadratically as for conventional RPA and hybrid functionals
- Scales cubically in system size (as DFT).

Prefactors are much larger than in DFT, but calculations for 200 atoms take less than 1 hour (128 cores)

Si defect calculations: 64-216 atoms

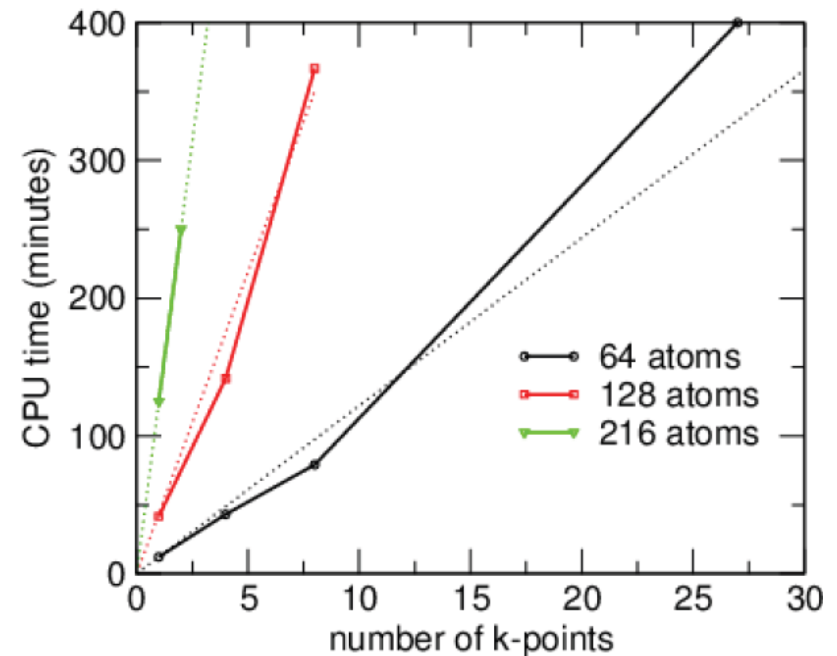
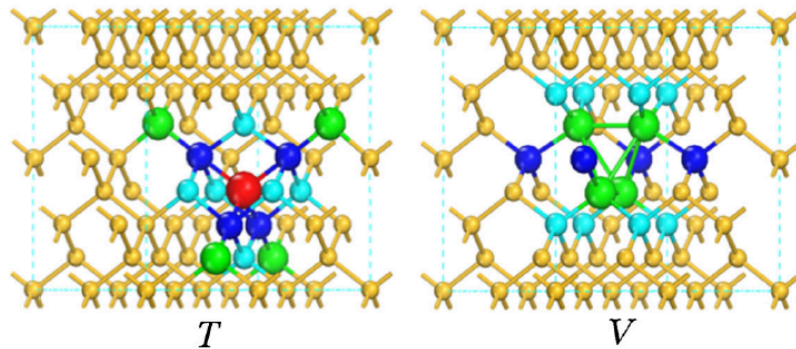
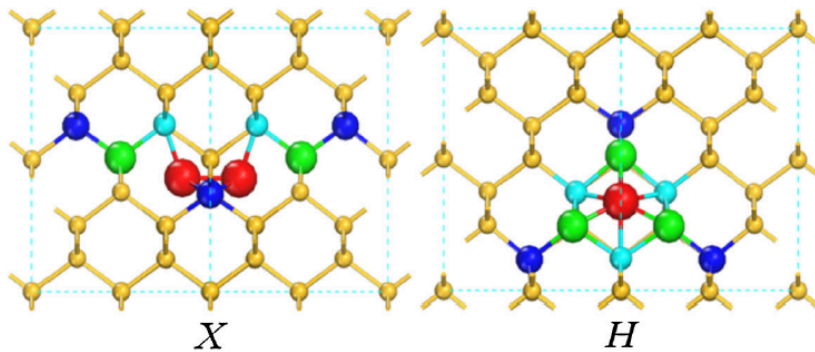


TABLE I. Timings in minutes for an RPA calculation for different bulk Si bcc cells. The calculations are done for the Γ point only and the number of cores is increased with system size. Since one of the computational steps scales only quadratically with system size, the total scaling is better than cubic.

Atoms	Cores	Time	Time \times cores / atoms ³ $\times 10^3$
54	32	14.3	2.91
128	64	83.2	2.54
250	128	299.9	2.45

Defect formation energies in Si

	PBE	HSE	HSE(+vdW)	QMC	RPA
Dumbbell X	3.56	4.43	4.41	4.4(1)	4.28
Hollow H	3.62	4.49	4.40	4.7(1)	4.44
Tetragonal T	3.79	4.74	4.51	5.1(1)	4.93
Vacancy	3.65	4.19	4.38		4.40

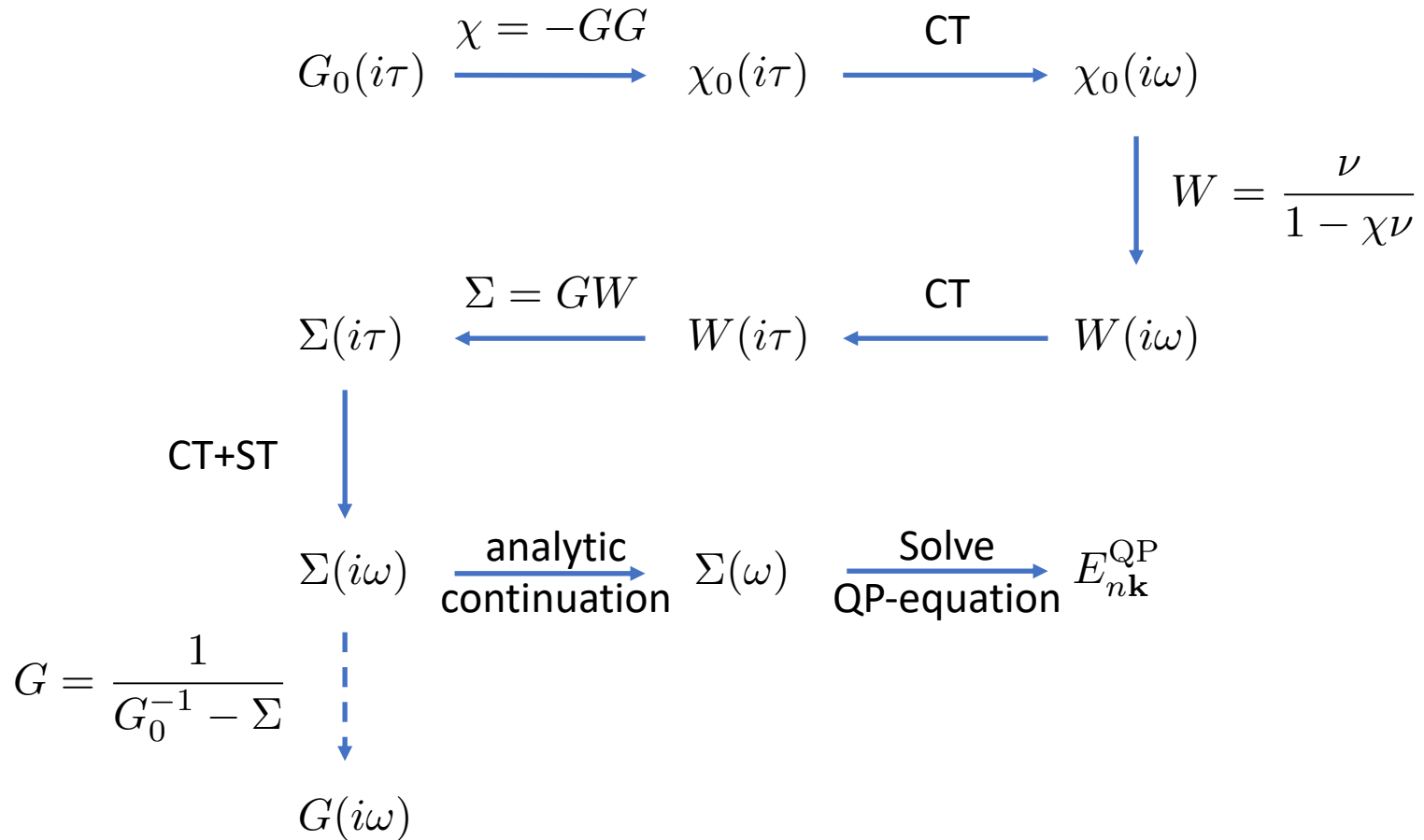


pictures and HSE+vdW:
Gao and Tkatchenko,
PRL 111, 045501 (2013).

QMC:
Parker, Wilkins, and Hennig,
Phys. Status Solidi B 248, 267 (2011).

Cubic-Scaling GW

P. Liu, M. Kaltak, J. Klimeš, and G. Kresse, PRB 94, 165109 (2016)



- Minimax grids:
M. Kaltak, J. Klimeš, and G. Kresse, JCTC 10, 2498 (2014); PRB 90, 054115 (2014).
- Cubic-scaling RPA total energies (ACFDT):
M. Kaltak, J. Klimeš, and G. Kresse, PRB 90, 054115 (2014).
- Cubic-scaling RPA quasi-particles (GW):
P. Liu, M. Kaltak, J. Klimeš, and G. Kresse, PRB 94, 15109 (2016).
- Finite temperature grids (!):
M. Kaltak and G. Kresse, PRB 101, 205145 (2020)

Dr. Merzuk Kaltak
(VASP Software GmbH)



Forces in the RPA

$$\frac{\partial E}{\partial R} = \langle \psi | \frac{\partial H}{\partial R} | \psi \rangle + \langle \frac{\partial \psi}{\partial R} | H | \psi \rangle + \langle \psi | H | \frac{\partial \psi}{\partial R} \rangle$$

Non-Hellmann-Feynman!

Hellmann-Feynman theorem: when the orbitals are eigenstates of the Hamiltonian, the second and third terms on the RHS are zero.

In the RPA we need to consider "non-Hellmann-Feynman" contributions!

These take the following form:

$$\langle \frac{\partial \psi}{\partial R} | H | \psi \rangle + \langle \psi | H | \frac{\partial \psi}{\partial R} \rangle \longrightarrow \left(\int G_0(i\omega) \frac{\partial E_{\text{RPA}}}{\partial G} \bigg|_{i\omega} G_0(i\omega) d\omega \right) \frac{\partial V_{\text{KS}}}{\partial R}$$

"from linear response"

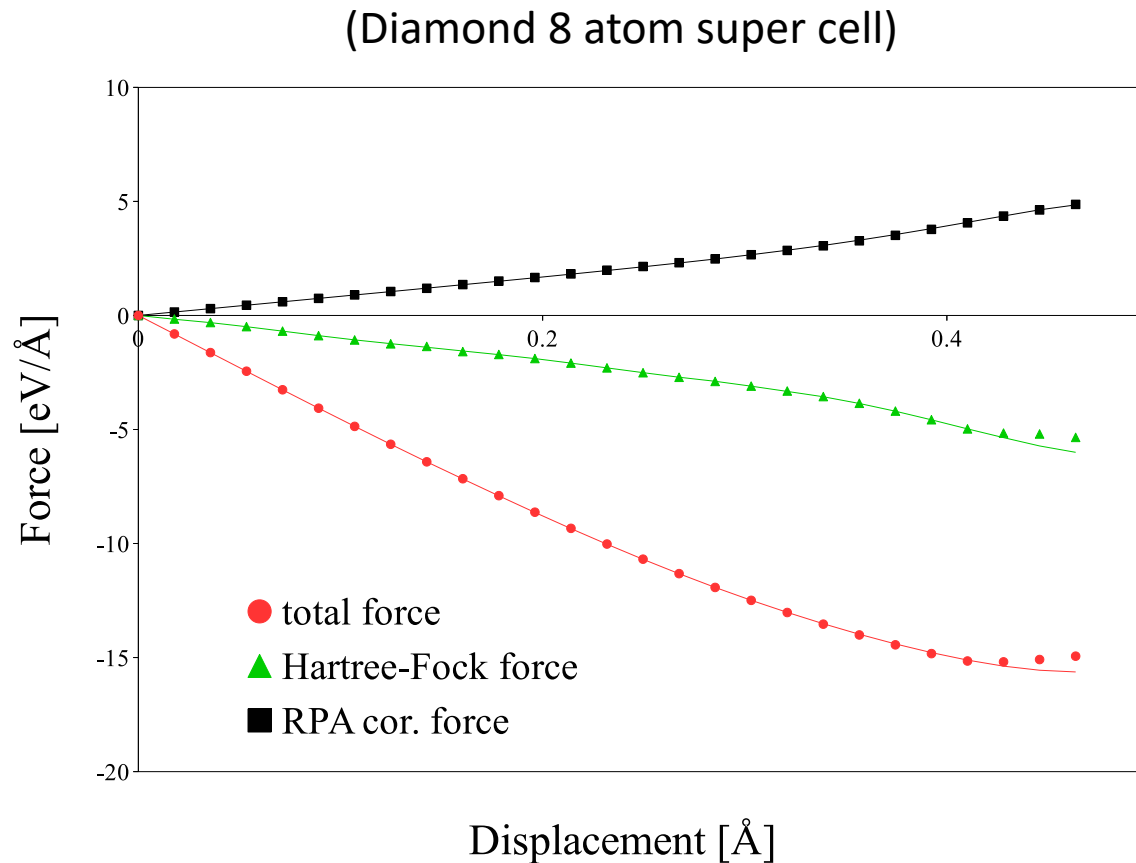
$$\frac{\partial E_{\text{H}}}{\partial G} = V_{\text{H}}$$

$$\frac{\partial E_{\text{X}}}{\partial G} = V_{\text{X}}$$

$$\frac{\partial E_{\text{c}}}{\partial G} = \Sigma$$

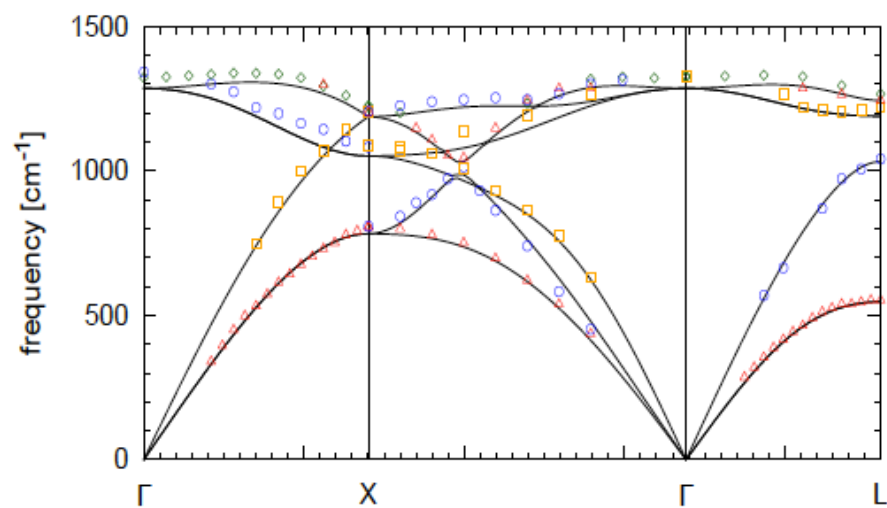
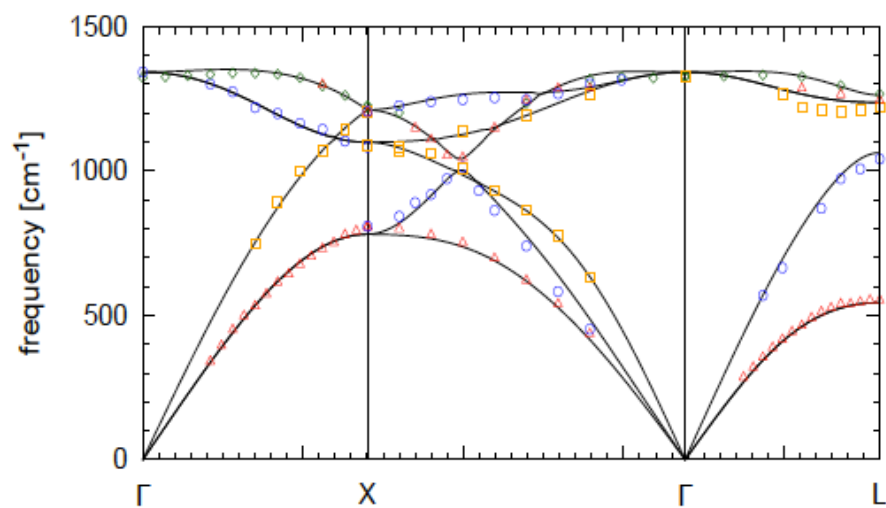
Forces in the RPA

A test: "RPA forces" (lines) vs. energy derivatives from finite differences (symbols):

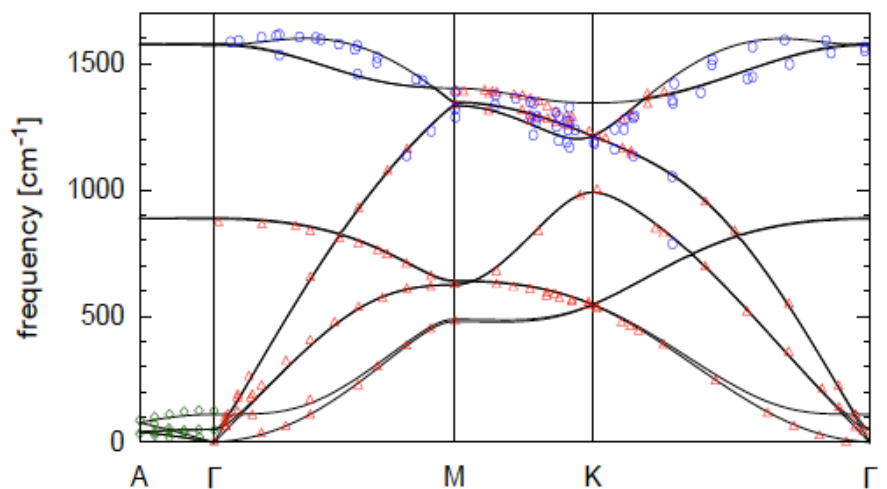
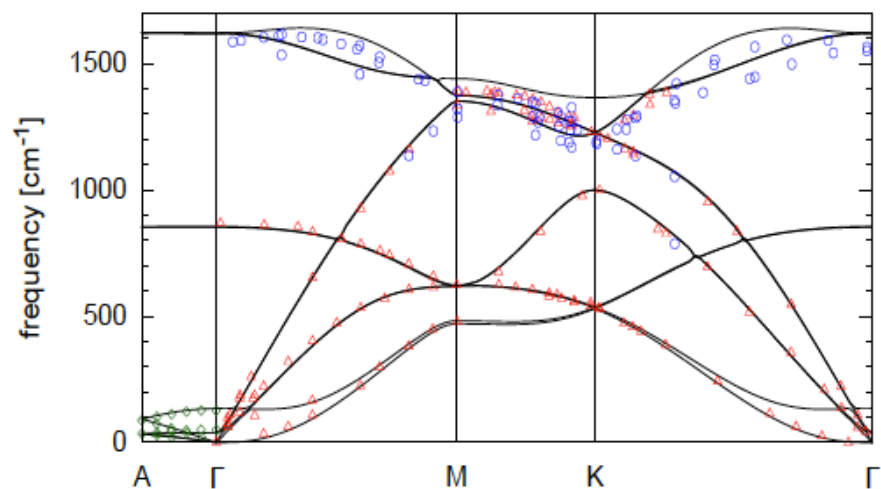


RPA

Diamond (128 atom super cell)

PBE**RPA**

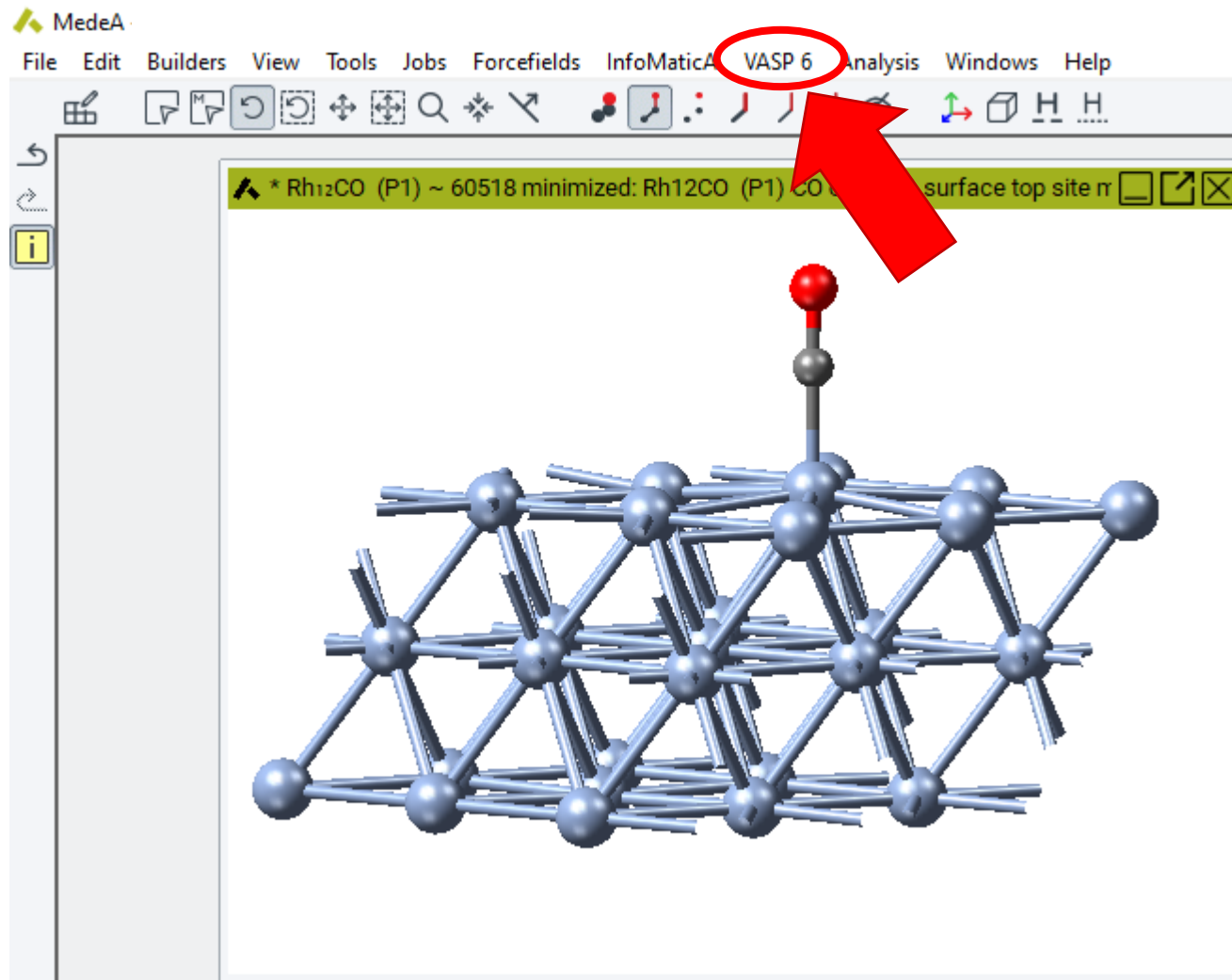
Graphite (128 atom super cell)

PBE

Summary

- Beyond DFT: the RPA
 - Well balanced description of all bond types (metallic, covalent, ionic, vdW)
 - Unfortunately energy differences can still not be predicted with chemical accuracy
 - But at the moment the optimal balance between accuracy and computational effort
- Cubic-Scaling RPA
 - Both for total energies as well as for quasi-particles
 - Minimax frequency and time grids
 - Near future: finite temperature RPA
- Forces in the RPA

Implementation in MedeA 3.1



Implementation in MedeA 3.1

MedeA : Run VASP 6

Calculation Functional/Potential SCF DOS/Optic/Tensors

Type of calculation Accurate Forces (Low Scaling ACFDT-RPA) ▼

Accurate Forces (Low Scaling ACFDT-RPA) Parameters

Calculate single point energy (all systems) ▼

Temperature mode finite temperature (all systems) ▼

Fermi smearing width: 0.1 eV

Number of frequency points: 12

Restore charge density full shape up to lmax = 4 ▼

Cutoff for response functions: eV

Maximum memory per MPI rank: MB

Implementation in MedeA 3.1

MedeA : Run VASP 6

Calculation Functional/Potential SCF DOS/Optic/Tensors

Type of calculation Accurate Forces (Low Scaling ACFDT-RPA) ▼

Accurate Forces (Low Scaling ACFDT-RPA) Parameters


Calculate single point energy (all systems) ▼

Temperature mode finite temperature (all systems) ▼

Fermi smearing width: 0.1 eV

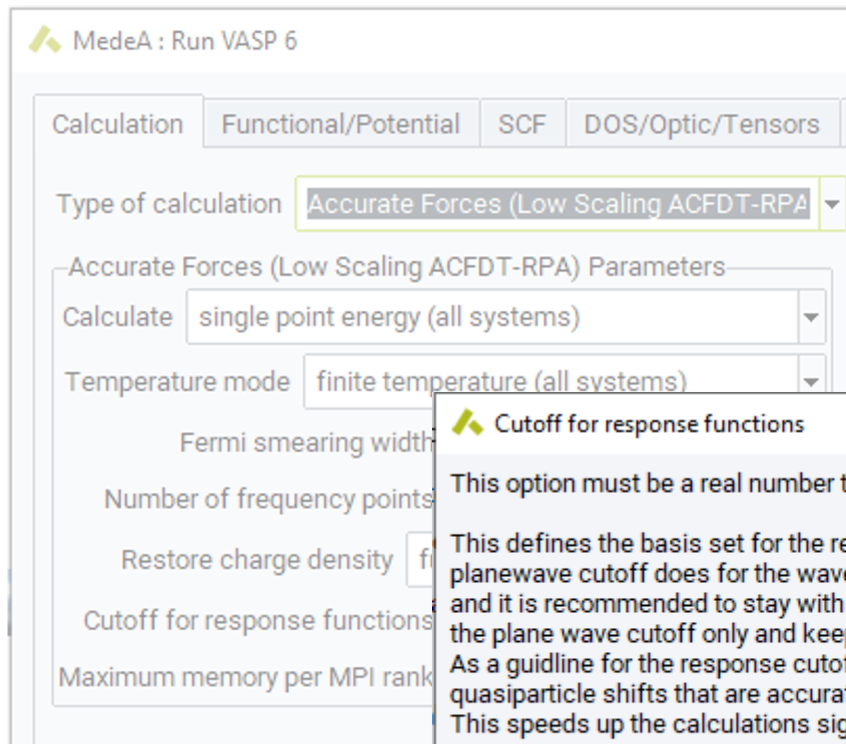
Number of frequency points: 12

Restore charge density full shape up to lmax = 4 ▼

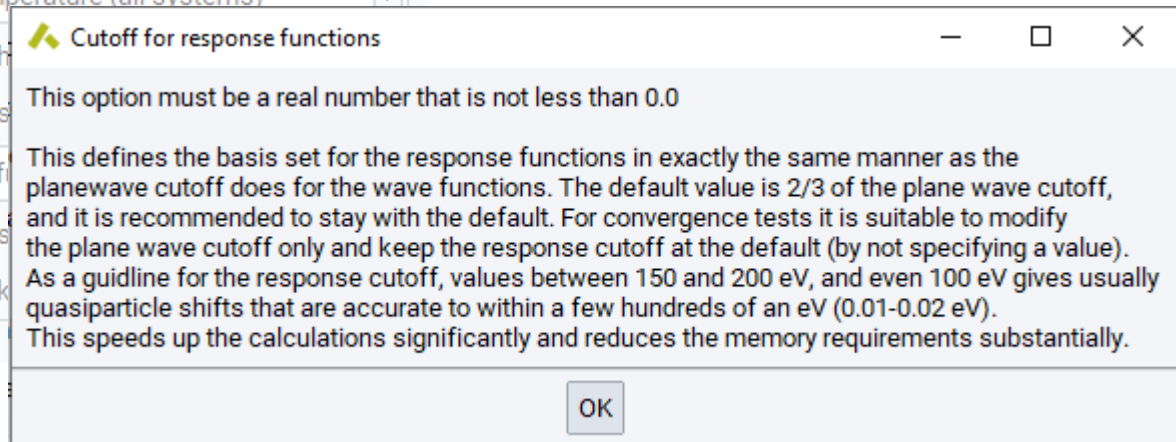
Cutoff for response functions:  eV

Maximum memory per MPI rank: MB

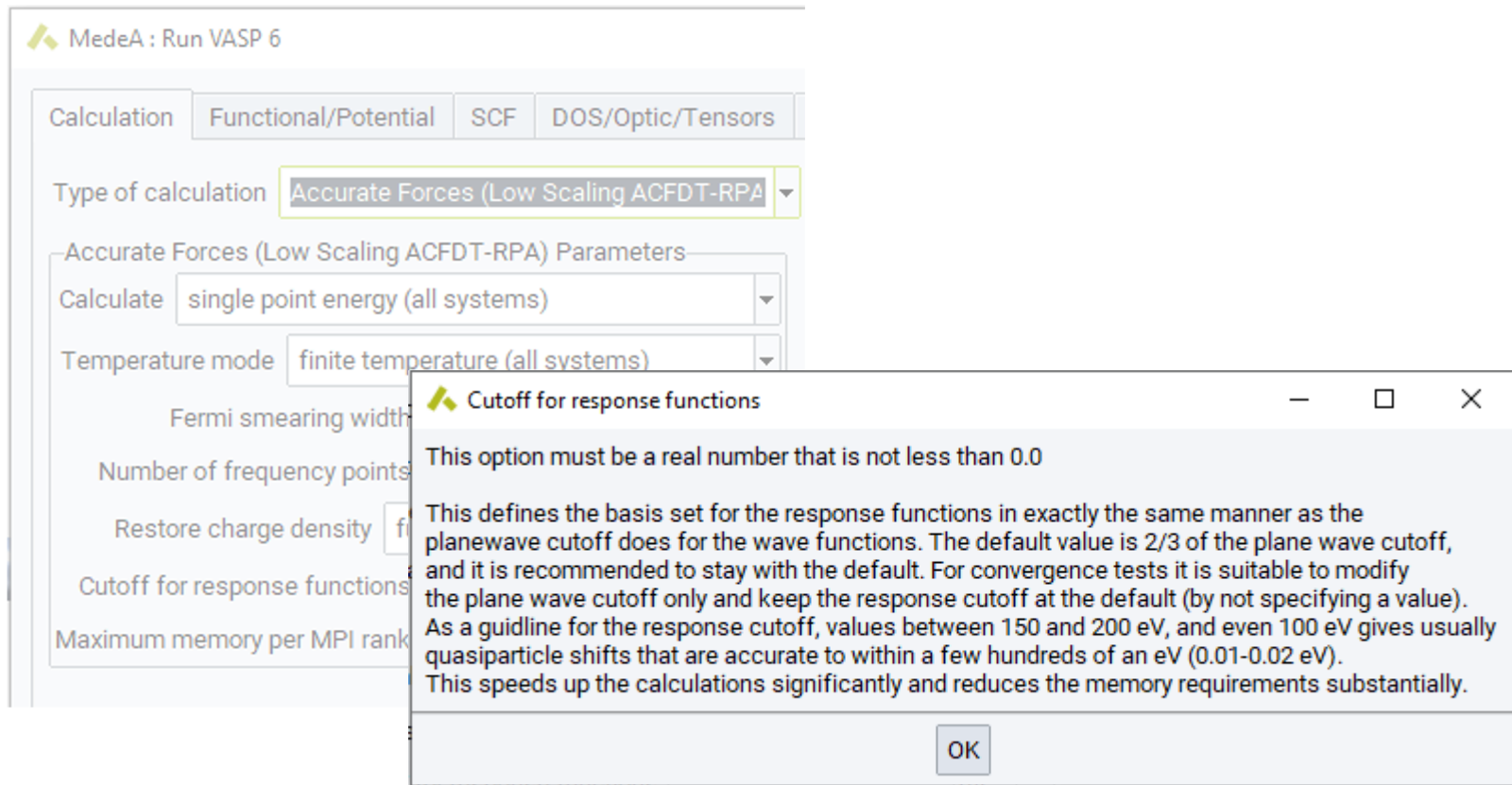
Implementation in MedeA 3.1



Context-sensitive help



Implementation in MedeA 3.1



More in the upcoming Webinars

The End

Thank you for your attention!

Materials Design UGM
San Diego, California
Oct. 13 – 15

www.materialsdesign.com/ugm-2020

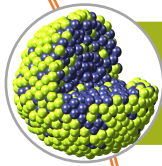




MedeA 3.1 Webinars



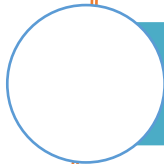
June 2020: Elasticity and beyond – Predicting Mechanical Properties with *MedeA*, by Ray Shan



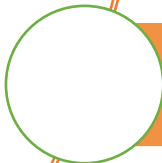
June 2020: Diffusion by Erich Wimmer and Benoît Minisini



July 2020: VASP 6 by Martjin Marsman



July 2020: Mesoscale modeling by Jörg-Rüdiger Hill



August 2020: *MedeA* 3.1 by Marianna, Walter and Jörg-Rüdiger Hill



Upcoming in the fall of 2020: Additional webinars addressing accuracy and larger scales



Questions about the webinar

Katherine Hollingsworth

khollingsworth@materialsdesign.com



materials design

info@materialsdesign.com

www.materialsdesign.com

MedeA

Innovation by Simulation

